



H2020-MSCA ITN  
Grant n. 956099



*Nan*ED



ISTITUTO ITALIANO  
DI TECNOLOGIA

# WP1 – 3D ED techniques

Mauro Gemmi – Istituto Italiano di Tecnologia



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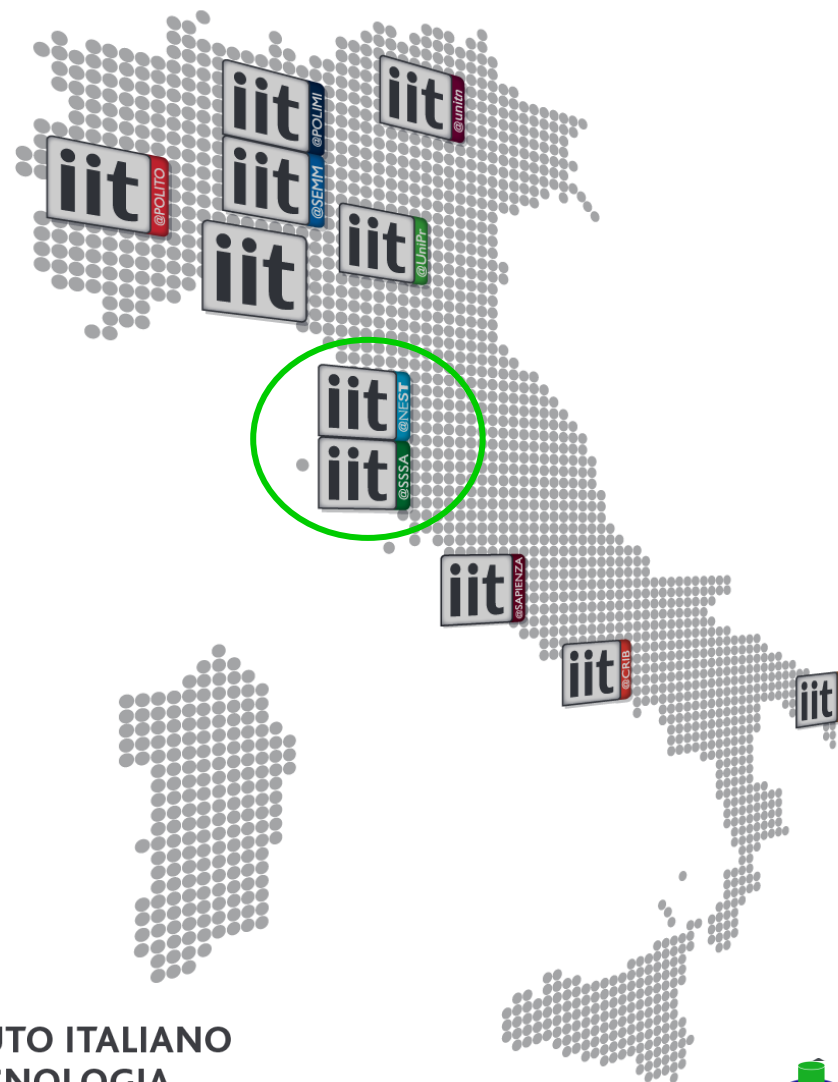
*“is intended to promote Italy's technological development and advanced education, consistent with national policies for scientific and technological development, thus strengthening the national production system”*



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CENTER FOR MATERIALS  
INTERFACES



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ELECTRON CRYSTALLOGRAPHY





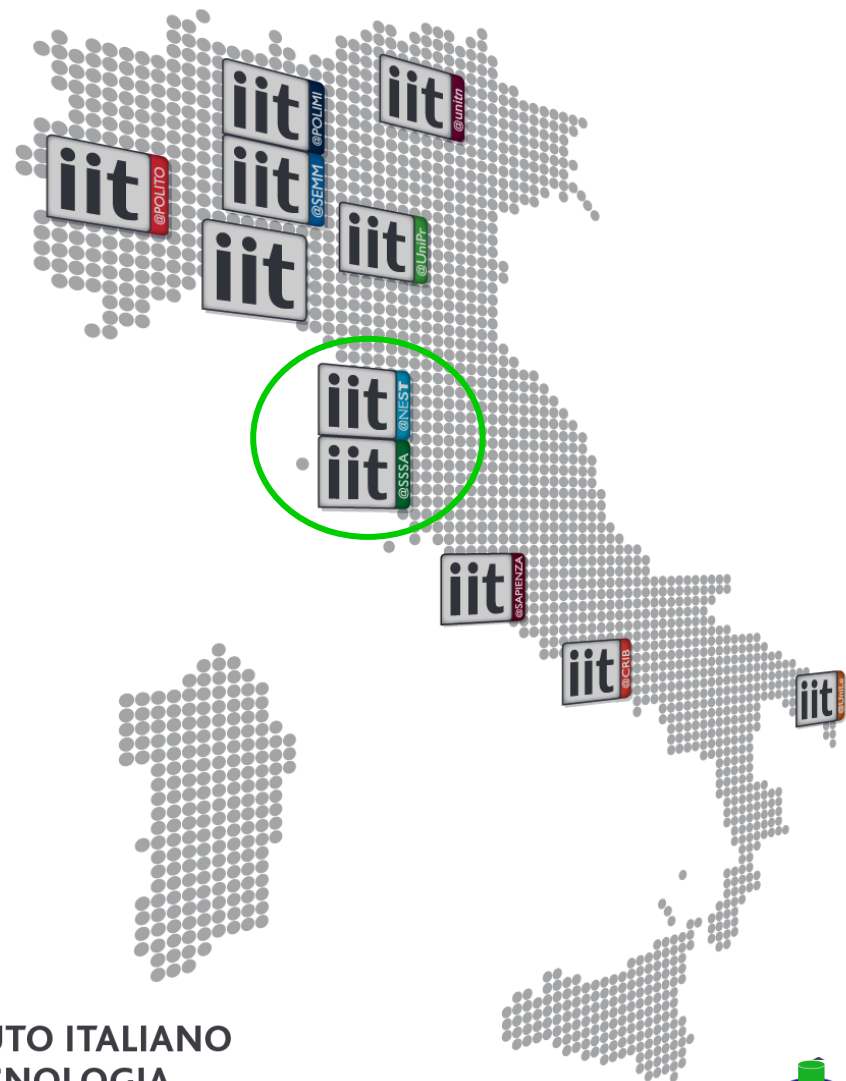
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INTERFACES

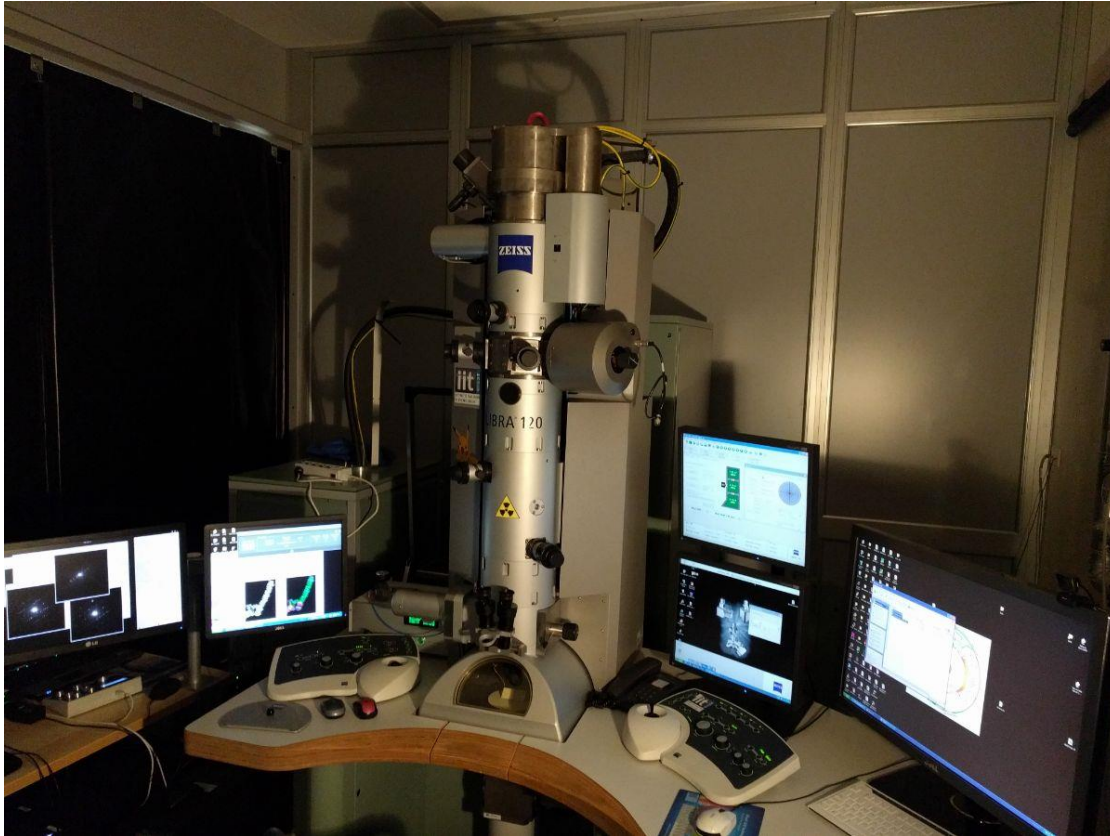


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ELECTRON CRYSTALLOGRAPHY





## INSTRUMENTS



Zeiss Libra 120 TEM

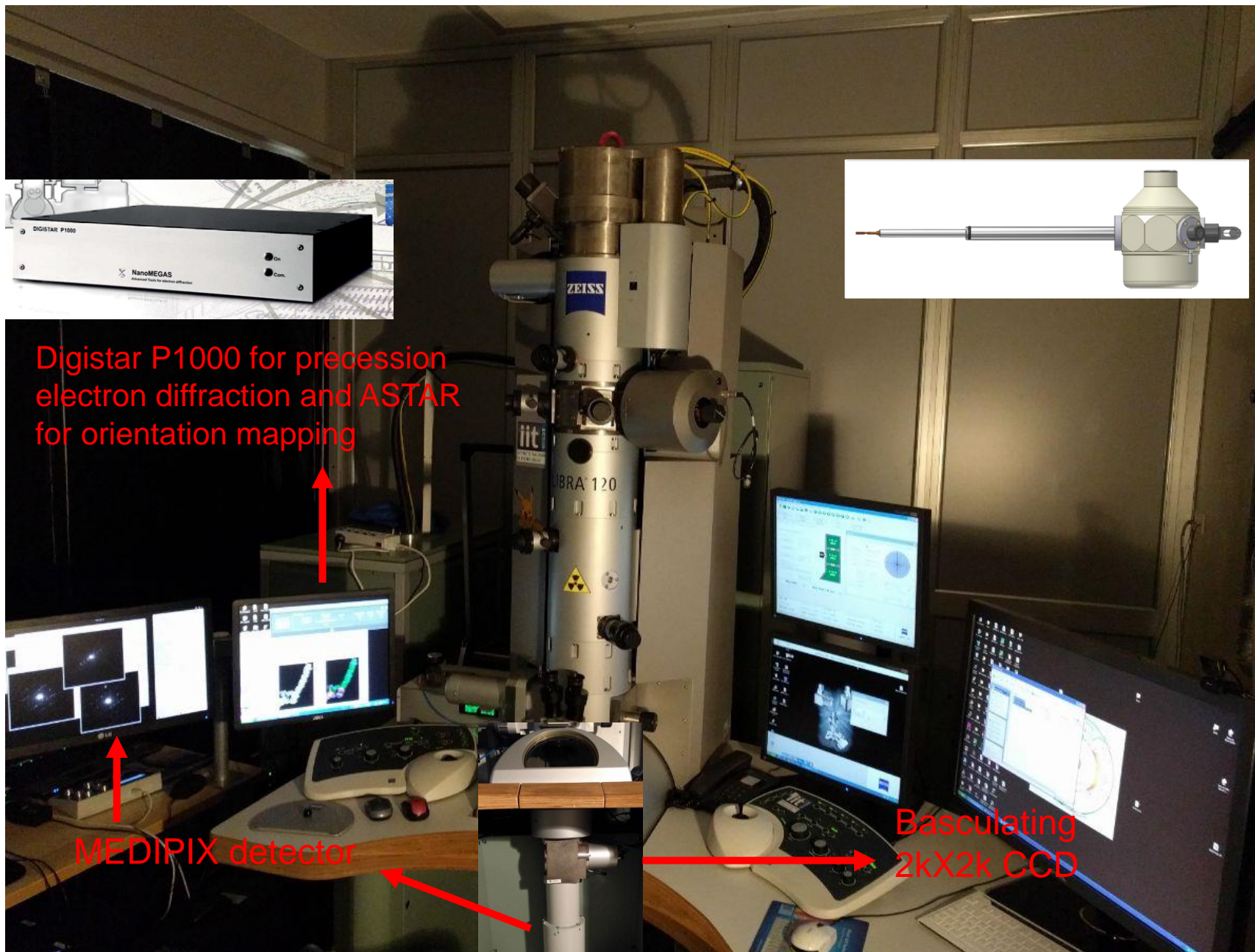
120 kV instrument

with in column energy filter



Stoe Stadi P diffractometer

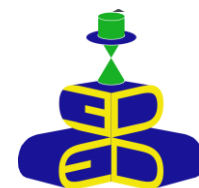




Digistar P1000 for precession  
electron diffraction and ASTAR  
for orientation mapping

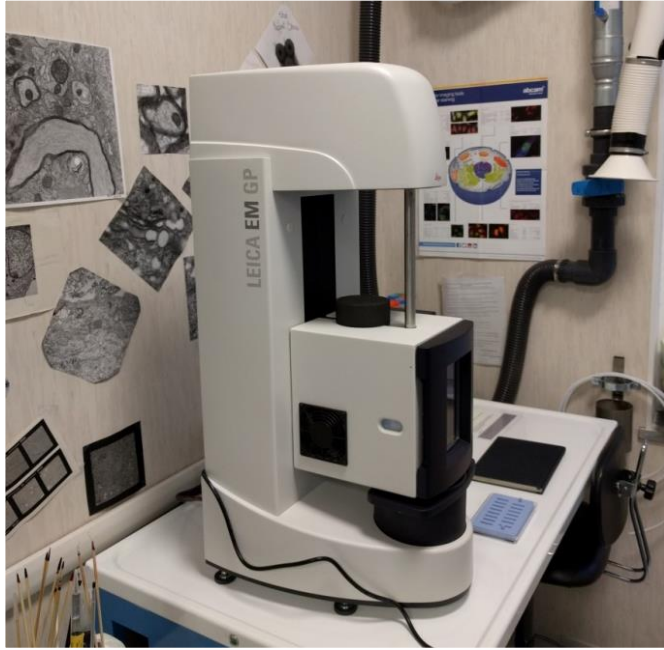
MEDIPIX detector

Basculating  
2kX2k CCD





# Sample Preparation



Leica Cryo plunging system



Leica UC7 Cryo ultramicrotome



# Synthesis Facility



Owen

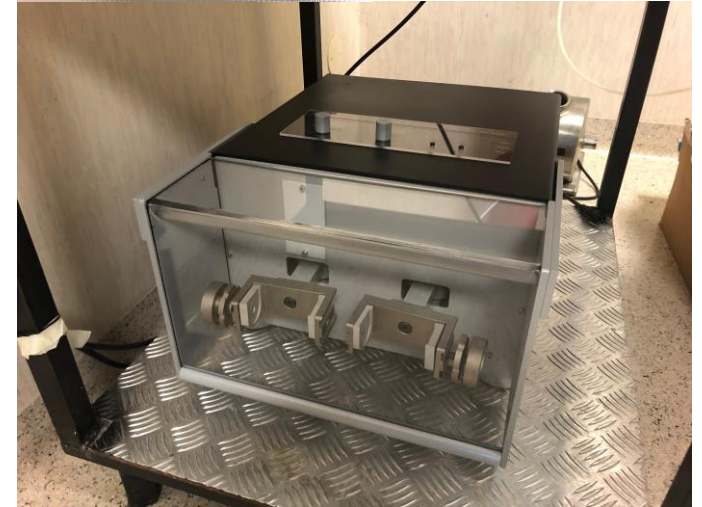
1 HT Owen maximum  $T=1650^{\circ}\text{C}$

1 medium T Owen  $T=650^{\circ}\text{C}$

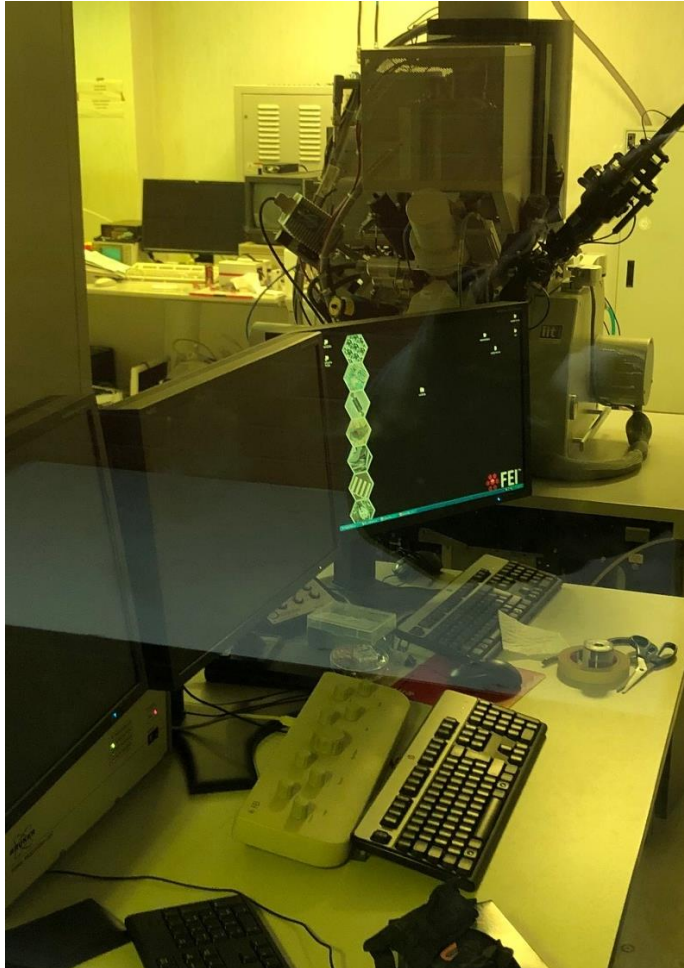
Shaker for Mechanochemical synthesis

Chemical synthesis facility

Chemical characterization facility







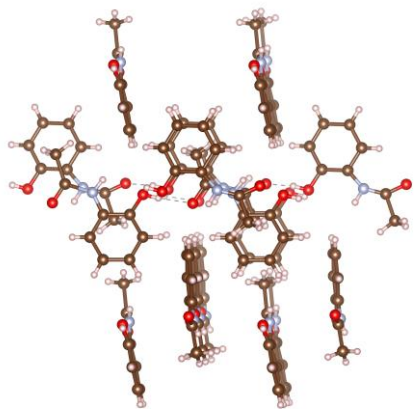
FIB ThermoFisher Helios 600



In the head quarter in Genova we will have access to a ThermoFisher Spectra 300 fully equipped.







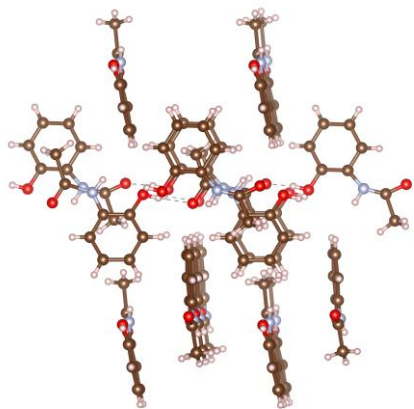
## WP1 Beam and vacuum sensitive materials

IIT

WP1 aims at the setting-up of sample preparation, data acquisition and data reduction strategies for an efficient structure characterisation of beam and vacuum sensitive materials.

- ☐ Fast and low dose data collection
- ☐ Fast crystal search strategies
- ☐ Special strategies of sample preparation to protect beam and vacuum sensitive samples

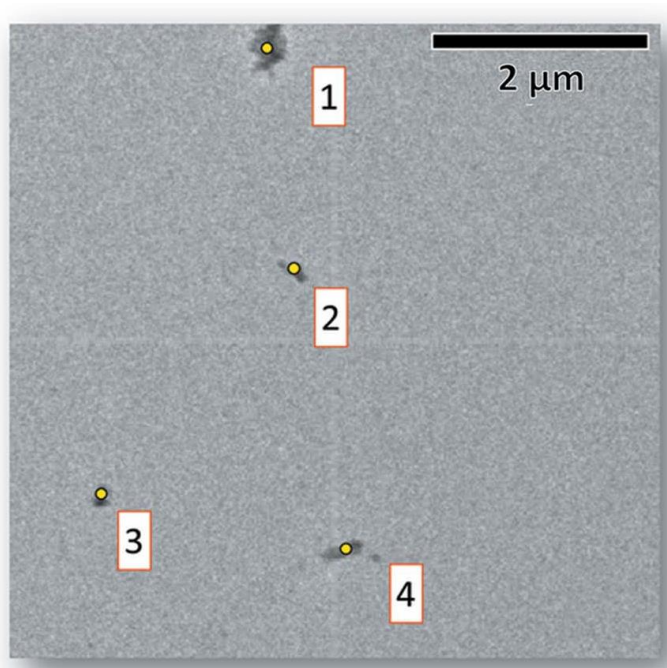




## WP1 Beam and vacuum sensitive materials

IIT

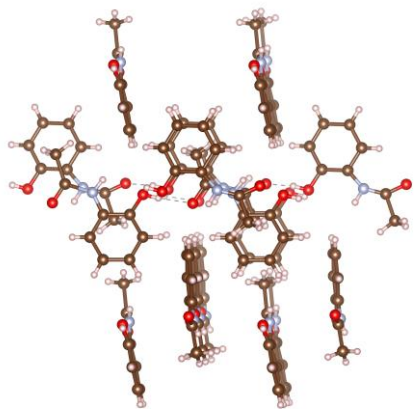
WP1 aims at the setting-up of sample preparation, data acquisition and data reduction strategies for an efficient structure characterisation of beam and vacuum sensitive materials.



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- ☐ Fast crystal search strategies
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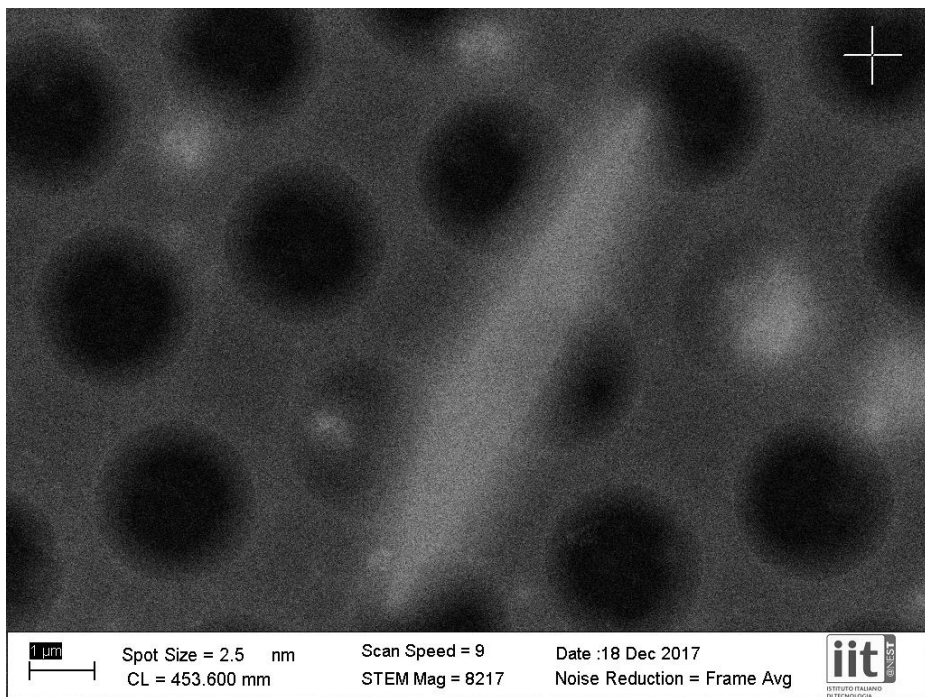




## WP1 Beam and vacuum sensitive materials

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WP1 aims at the setting-up of sample preparation, data acquisition and data reduction strategies for an efficient structure characterisation of beam and vacuum sensitive materials.



- ☐ Fast and low dose data collection
- ☐ Fast crystal search strategies
- ☐ Special strategy of sample preparation to protect beam and vacuum sensitive samples



## WP1 Beam and vacuum sensitive materials

T1.1: Low-dose data collection protocols (**ESR1/IIT, ESR2/IIT, ESR8/JGU, ESR11/SU**)

T1.2: Reducing vacuum and beam sensitivity through sample preparation (**ESR1/IIT, ESR8/JGU**)

T1.3: Structure solution, polymorphism determination and high throughput data collection for pharmaceutical and other beam sensitive compounds (**ESR1/IIT, ESR2/IIT, ESR11/SU**)

T1.4: Standardization of 3D ED data collection protocols (**All ESR**)



In collaboration with IUCr we need to create specific entries for structure solved and refined with 3D ED data that will be inserted in the CIF vocabulary

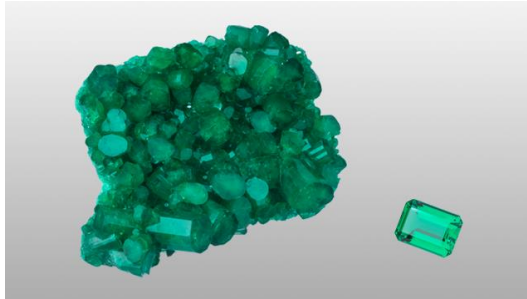
The final goal is a CIF fully compatible and able to store all the details of data collection and analysis in case of 3D ED data.





# What is all about?

Natural or synthetic  
unknown crystal

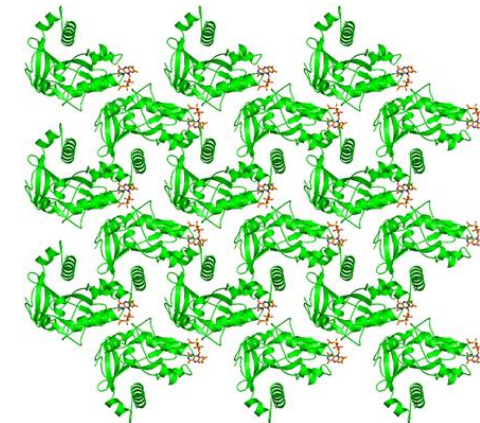
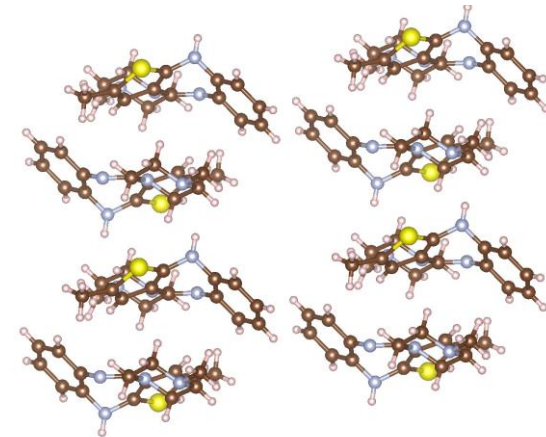
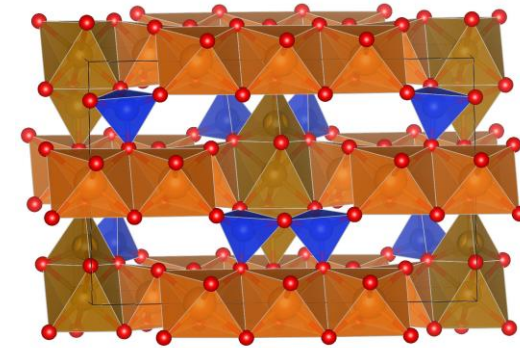


## THE CRYSTALLOGRAPHIC PROBLEM



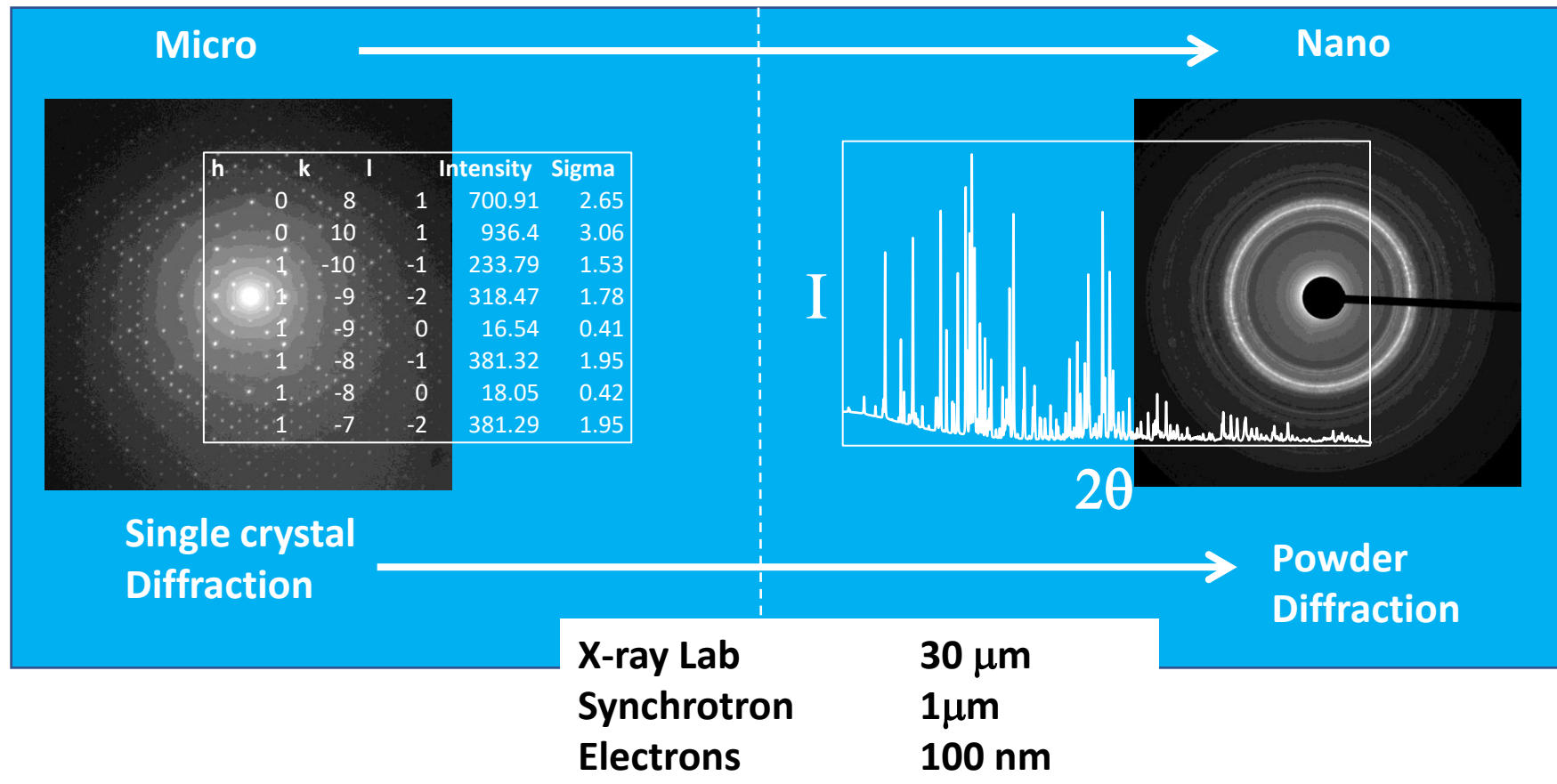
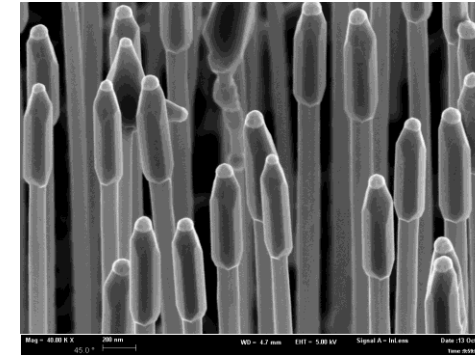
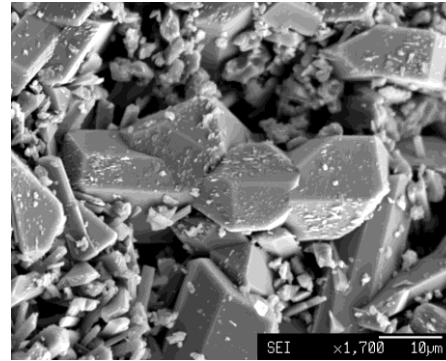
Having a minimal knowledge:  
Tentative chemical composition  
Maybe we have the molecule.....

**When the crystals get nano!!!!**



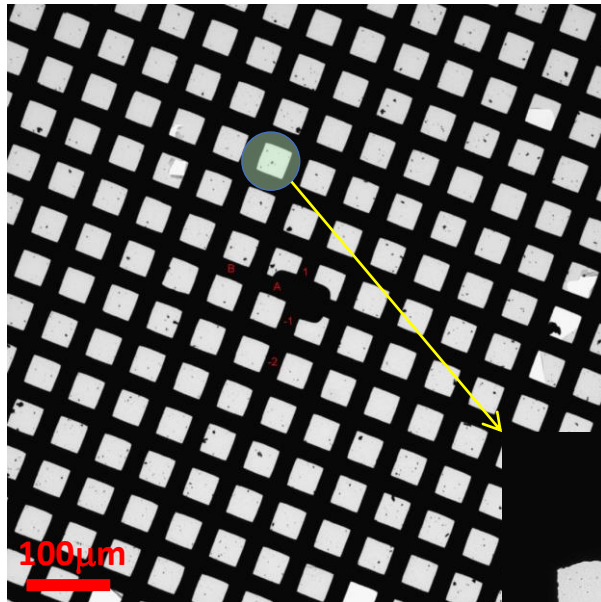
# ELECTRON DIFFRACTION FOR INVESTIGATING A CRYSTAL STRUCTURE

S  
I  
Z  
E

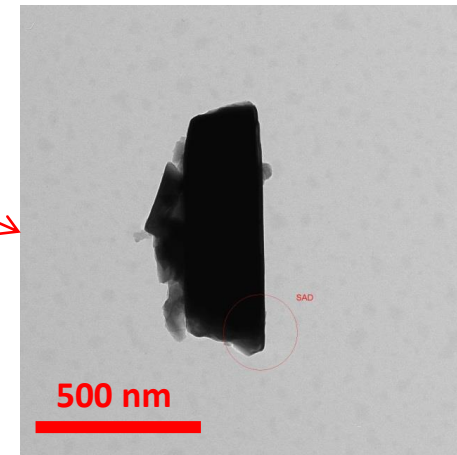
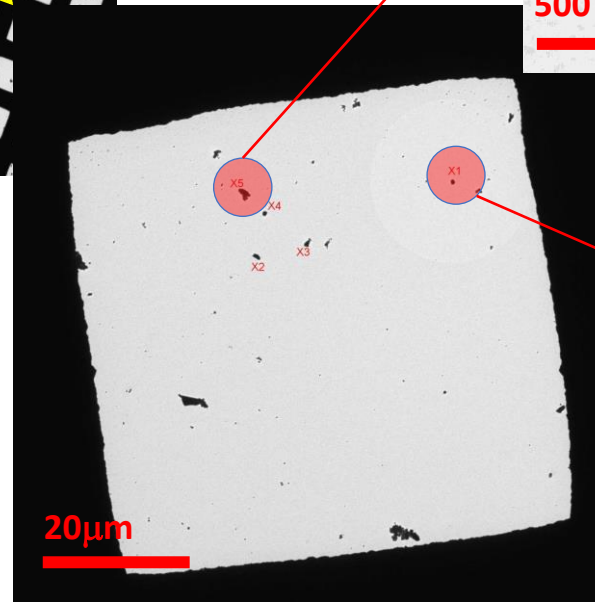
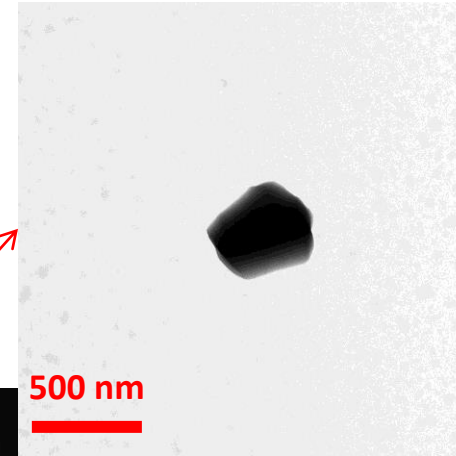




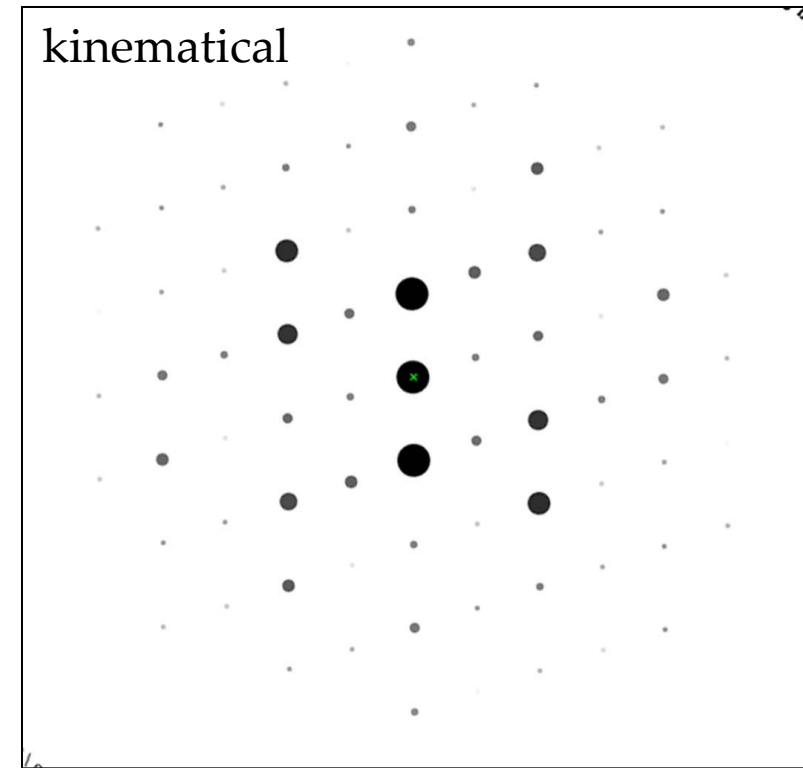
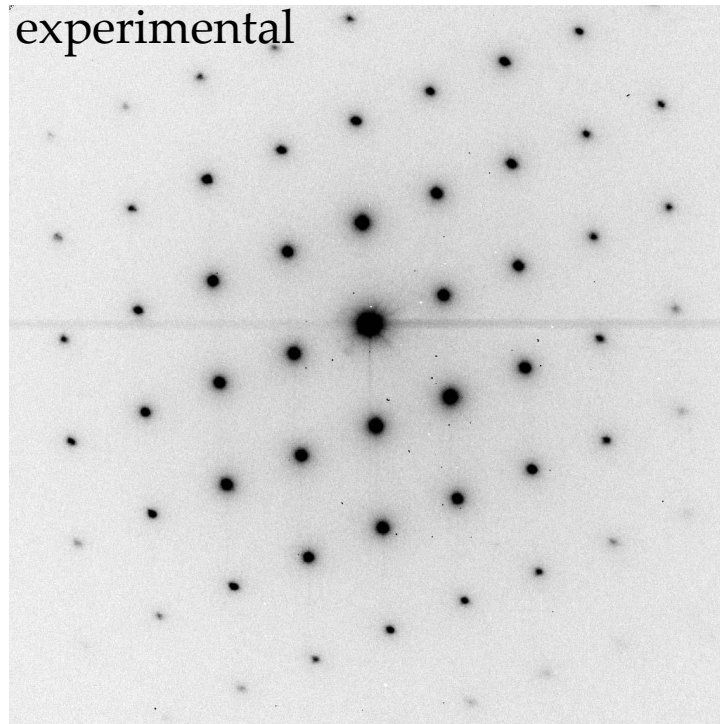
In a tem we can analyse crystal by crystal.....minimum size  
for an ADT exp is in the range of 100nm



We have a single  
crystal  
diffractometer for  
nanometric crystals



## DYNAMICAL EFFECTS (NO FIRST ORDER BORN APPROX.)



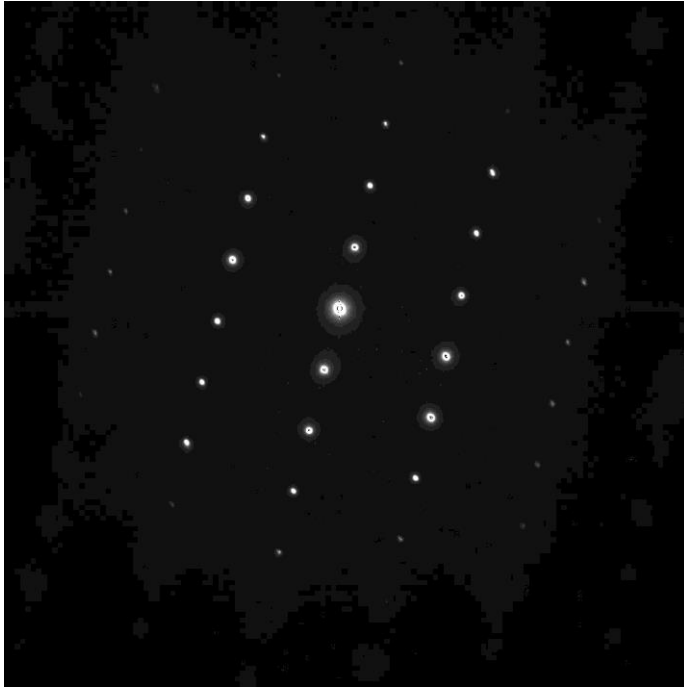
[100] Calcite ( $\text{CaCO}_3$ )

- distribution of intensities completely altered
- the reflections intensities become very similar

$$I(hkl) = |\text{Fourier}(V(\vec{r}))|^2$$







Oriented Zone axis

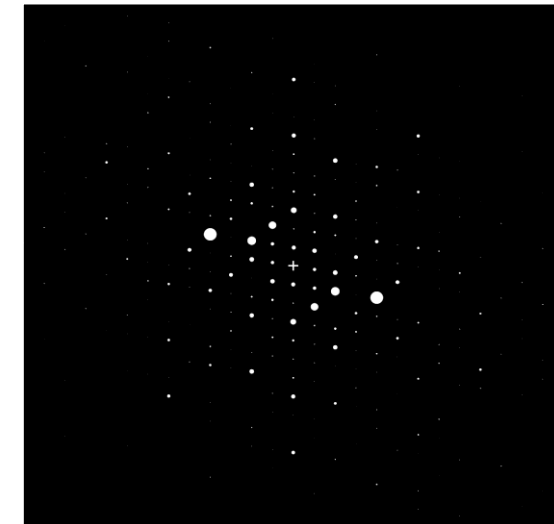
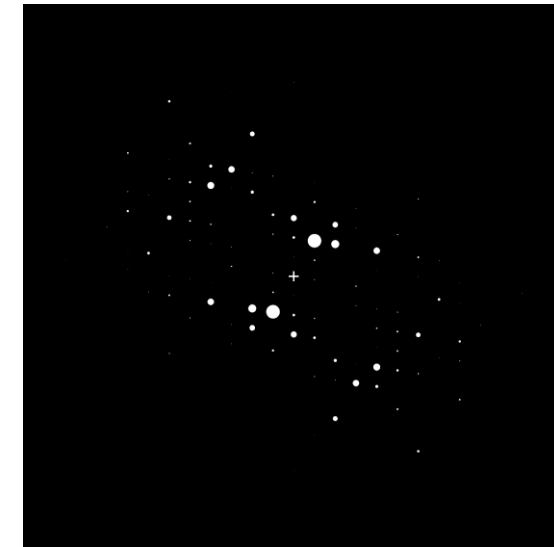
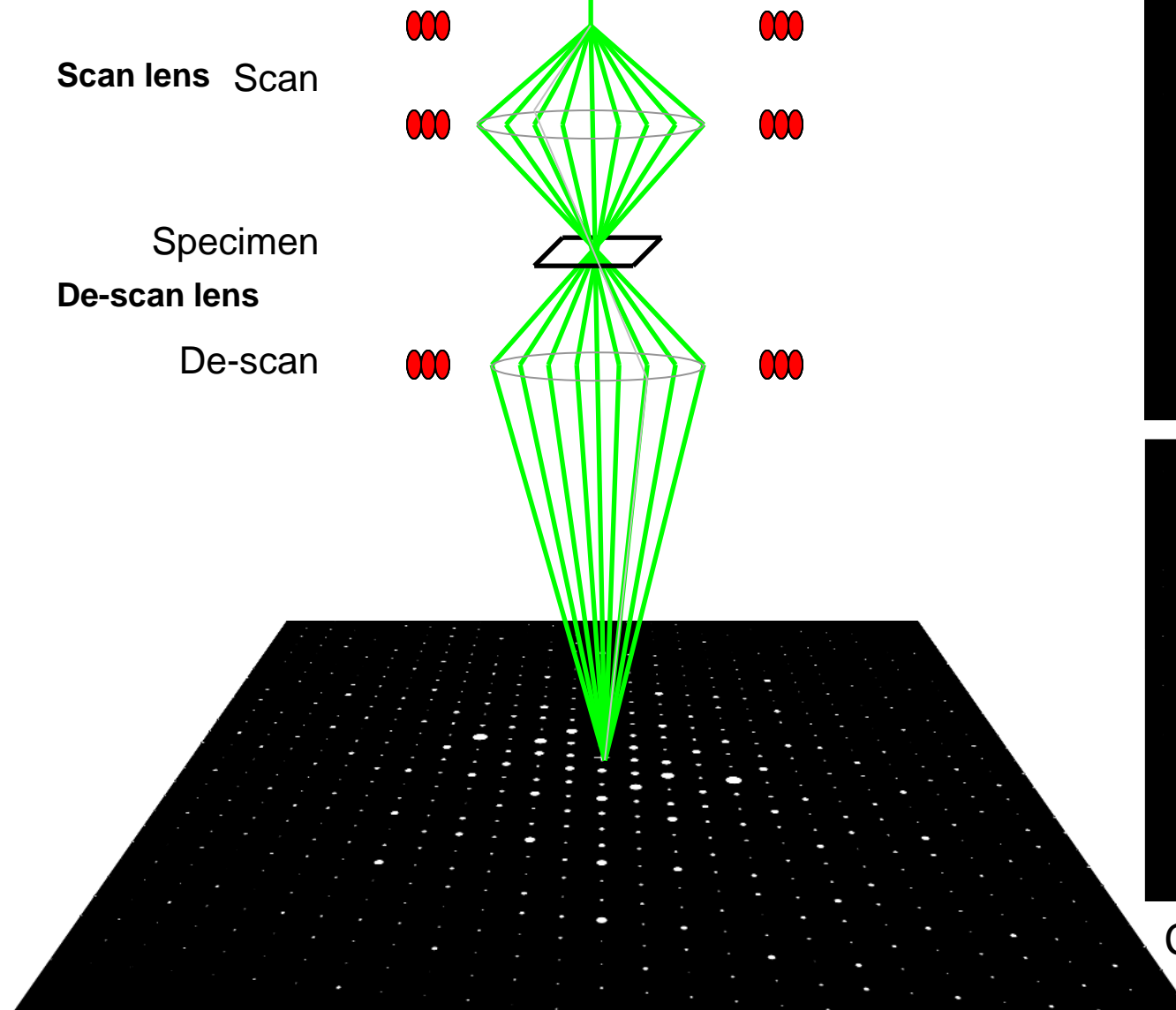


Not oriented



# Precession electron diffraction

Reference : C.Own PhD thesis



Conventional Diffraction Pattern

Vincent and Midgley Ultramicroscopy 53 (1994) 271.

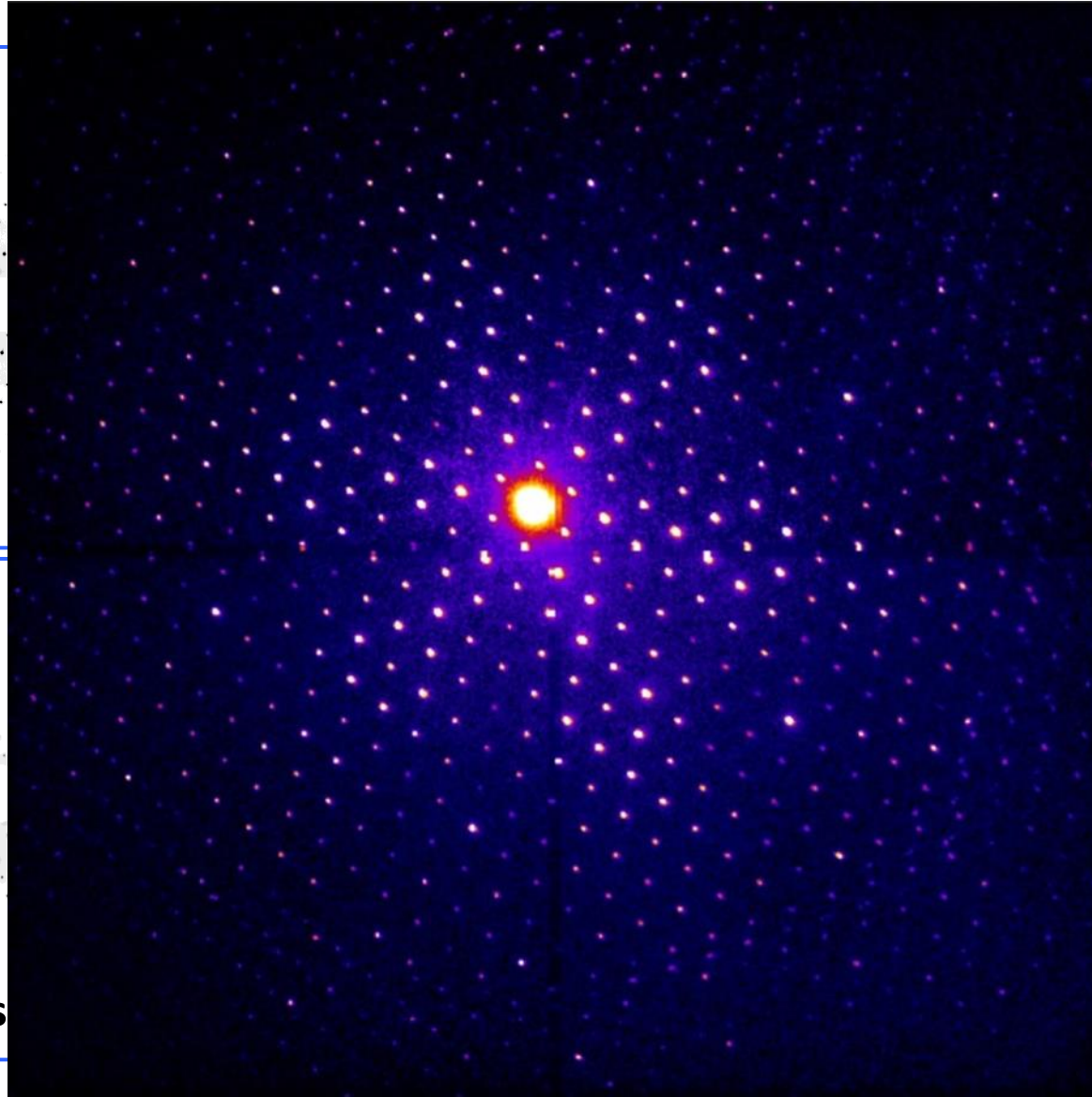


## FROM STANDARD SAED TO PED

[111]

[100]

Precess

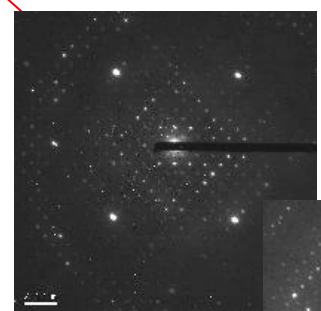


ray like)

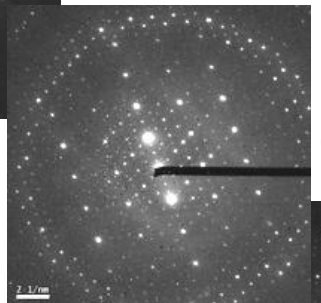




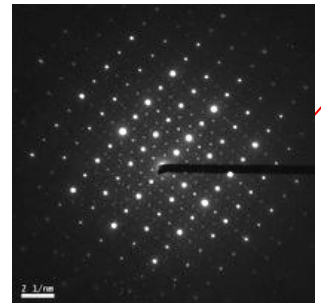
# ZONE AXIS PATTERNS



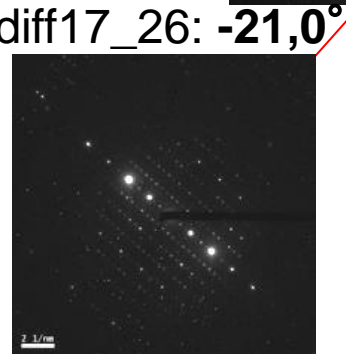
diff18\_25: **28,5°**



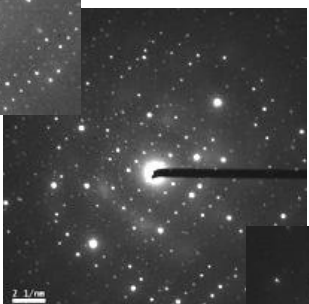
diff21\_26: **22,2°**



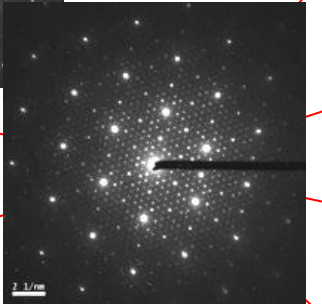
diff20\_26: **-33,5°**



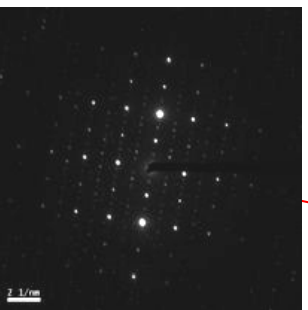
diff17\_26: **-21,0°**



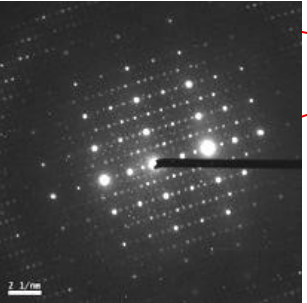
diff18\_26: **15,1°**



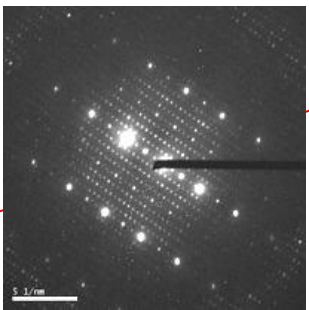
diff19\_26: **0°**



diff2\_27: **-11,6°**



diff16\_26: **19,5°**

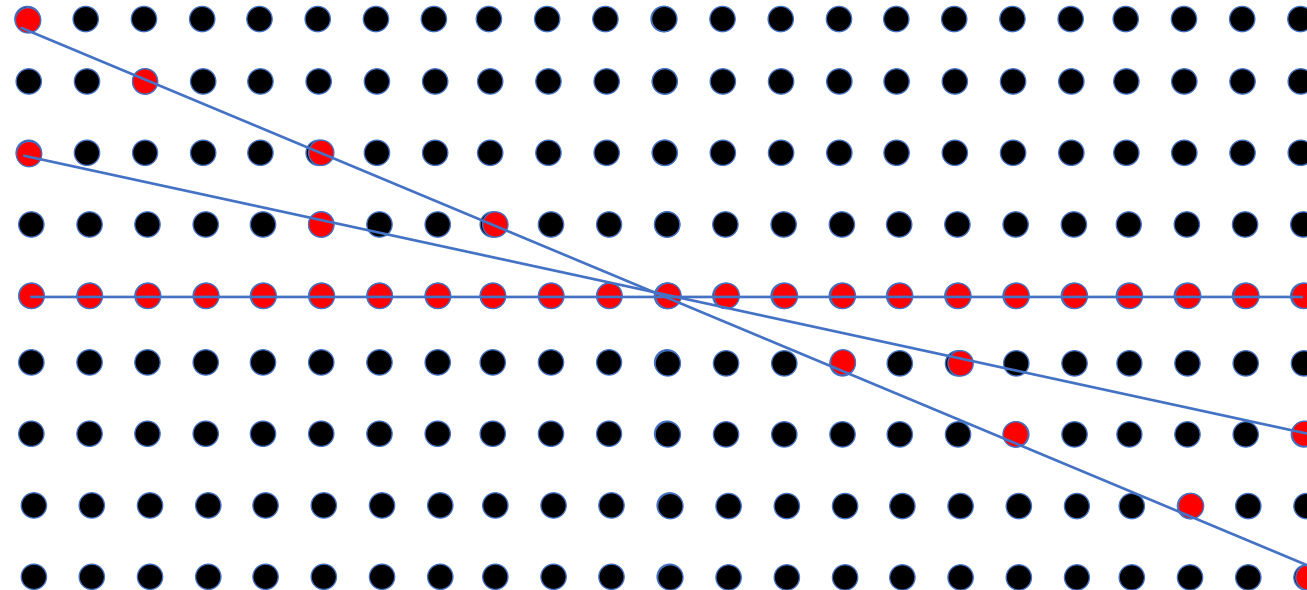


diff3\_26: **29,0°**

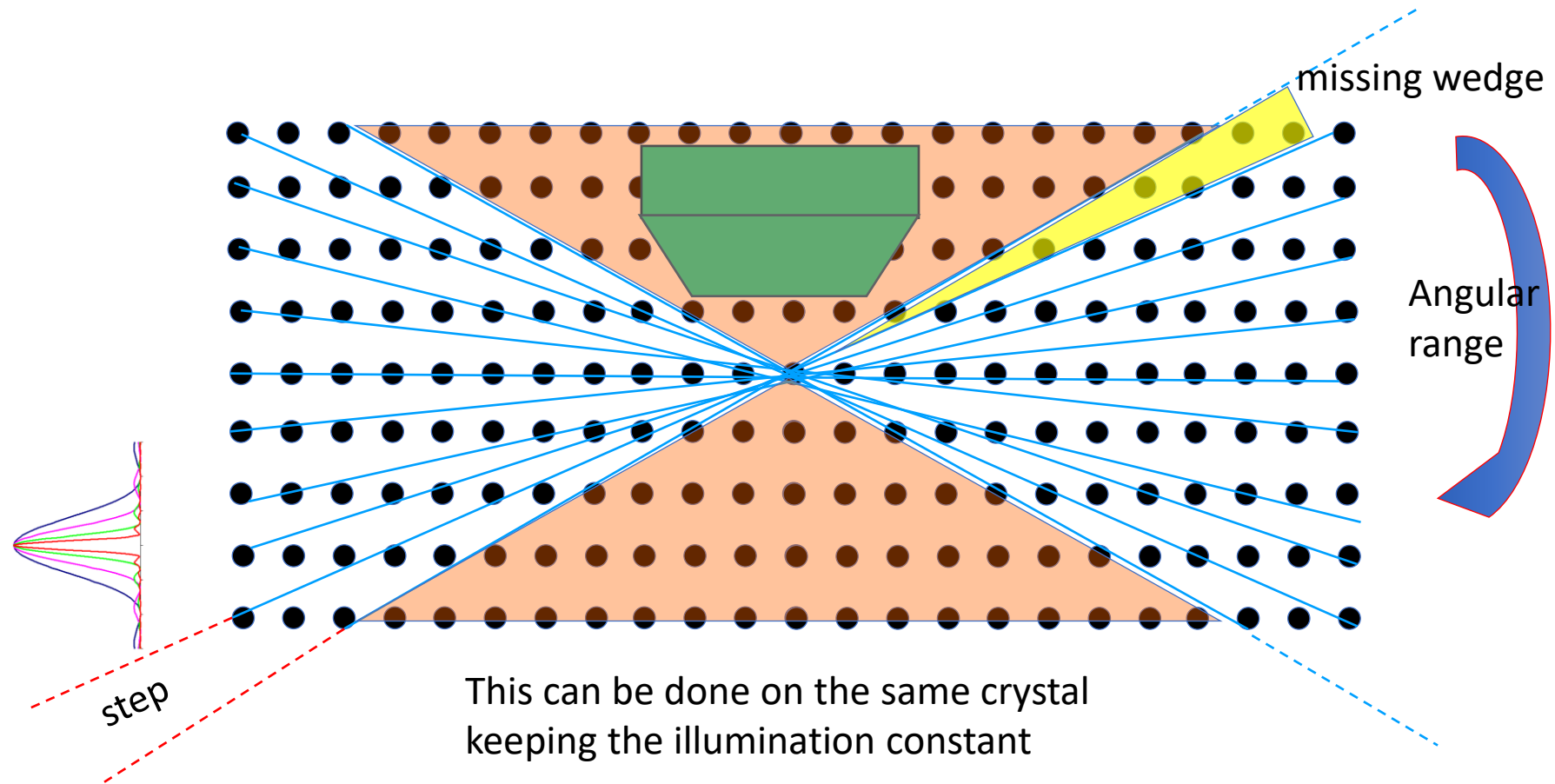
Limitations:  
Reciprocal space coverage  
Rescaling



## HOW TO SCAN THE RECIPROCAL SPACE



# ELECTRON DIFFRACTION TOMOGRAPHY



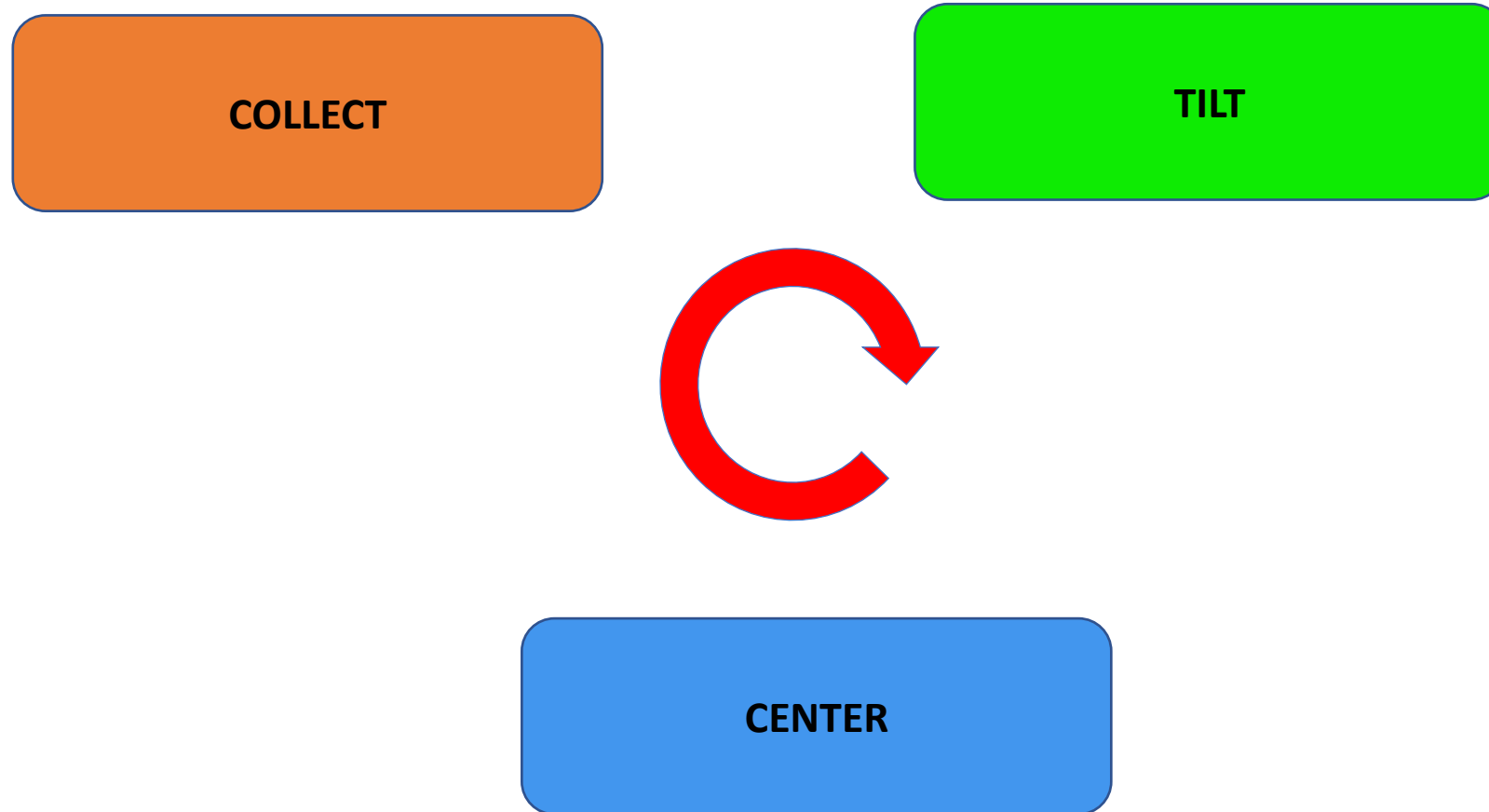
Dynamical effect ok  
Excitation error to be solved

Angular range to maximize  
Crystal moves





# EDT FLOW CHART



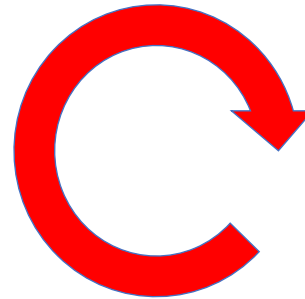
# EDT FLOW CHART

## COLLECT

Collect a diffraction Pattern	SAED	Precession mode
	Nano diffraction	Stationary beam

## TILT

Tilt the crystal	Mechanical	1°-2°
	Electrical	0.01°

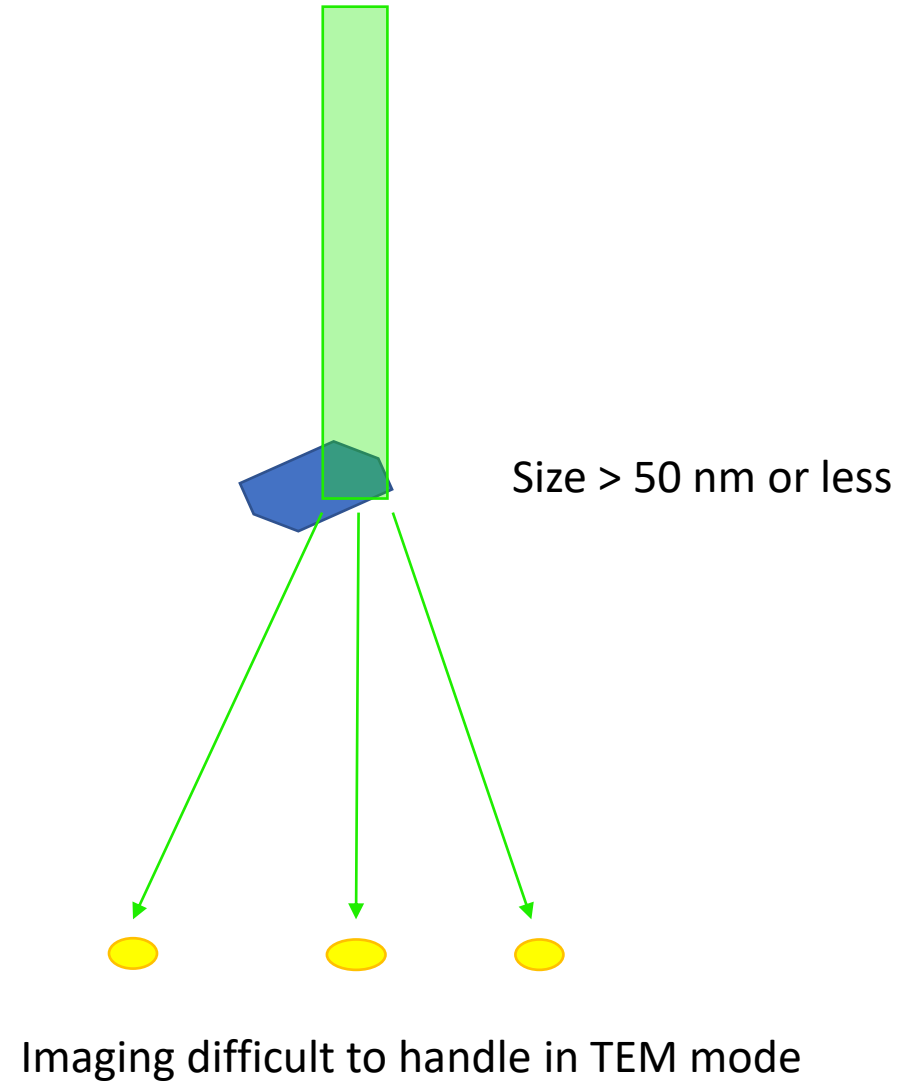
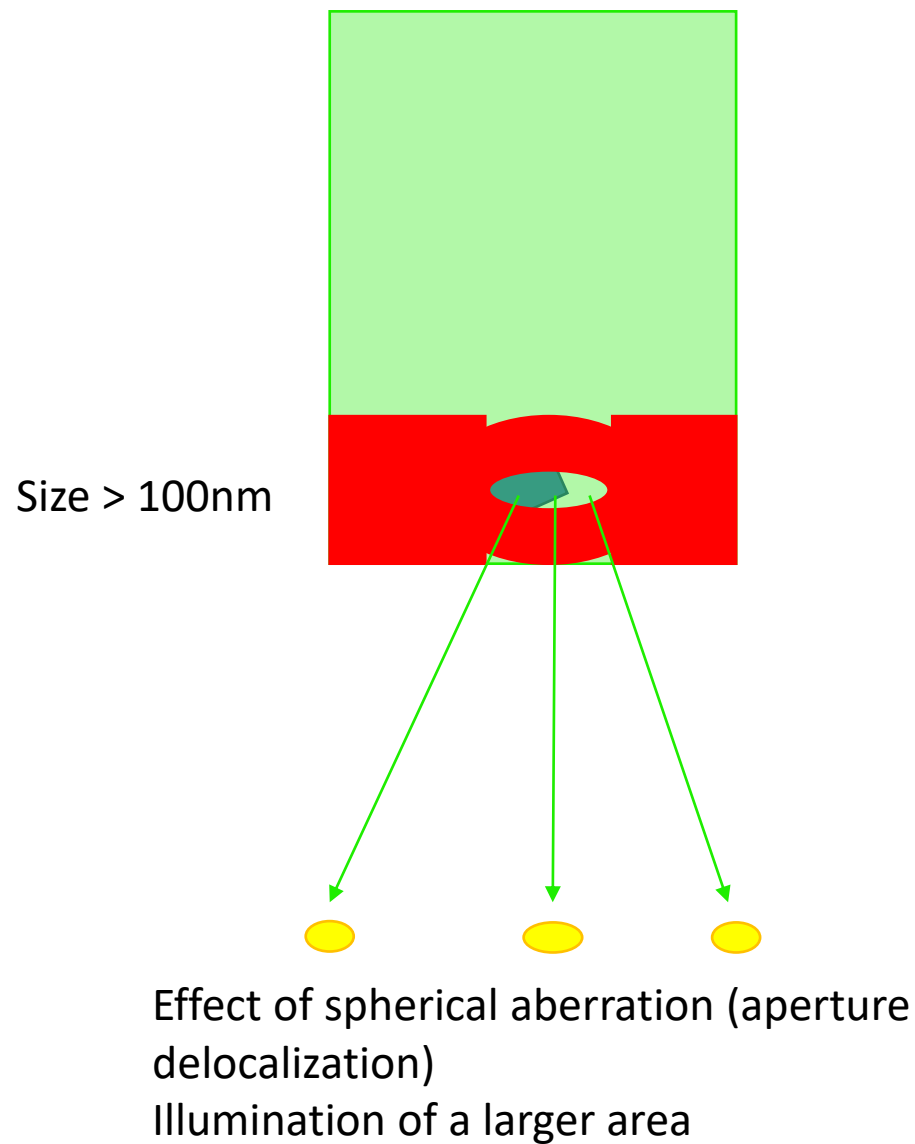


## CENTER

Center the crystal	Manually	TEM
	Automatically	STEM

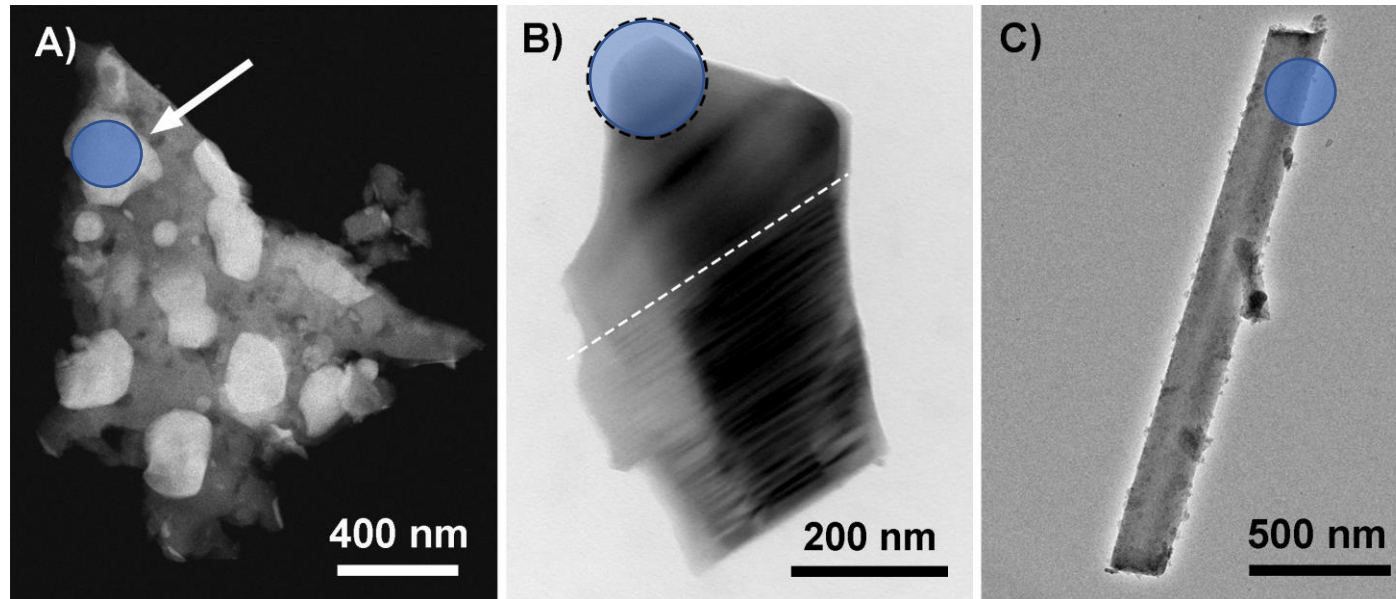


# SAED vs. NANODIFFRACTION





## 3D NANODIFFRACTION

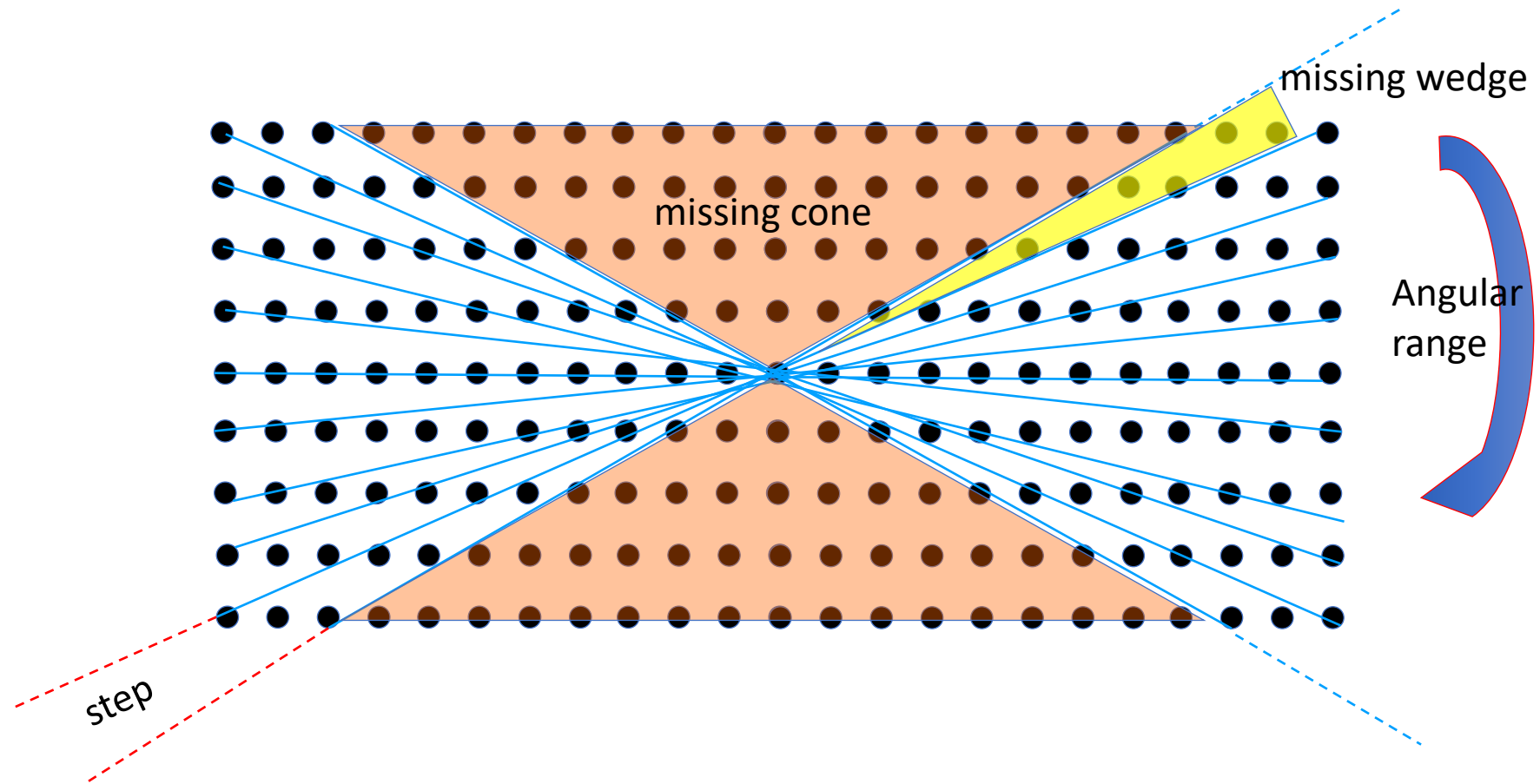


Local crystallography

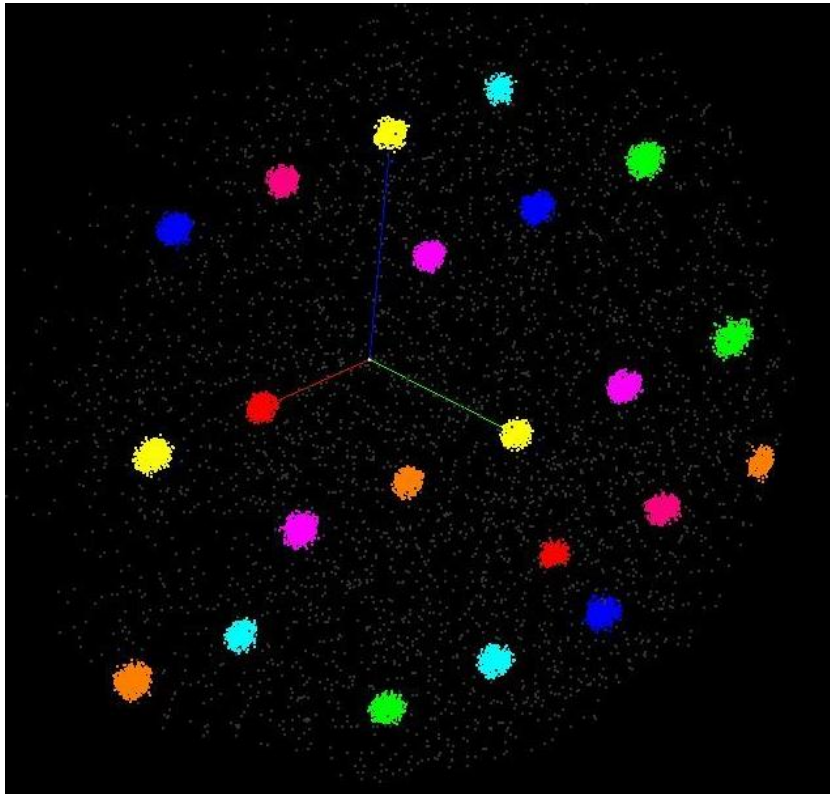
Intergrowth – Twinning – Order disorder – Polytipism



# ELECTRON DIFFRACTION TOMOGRAPHY

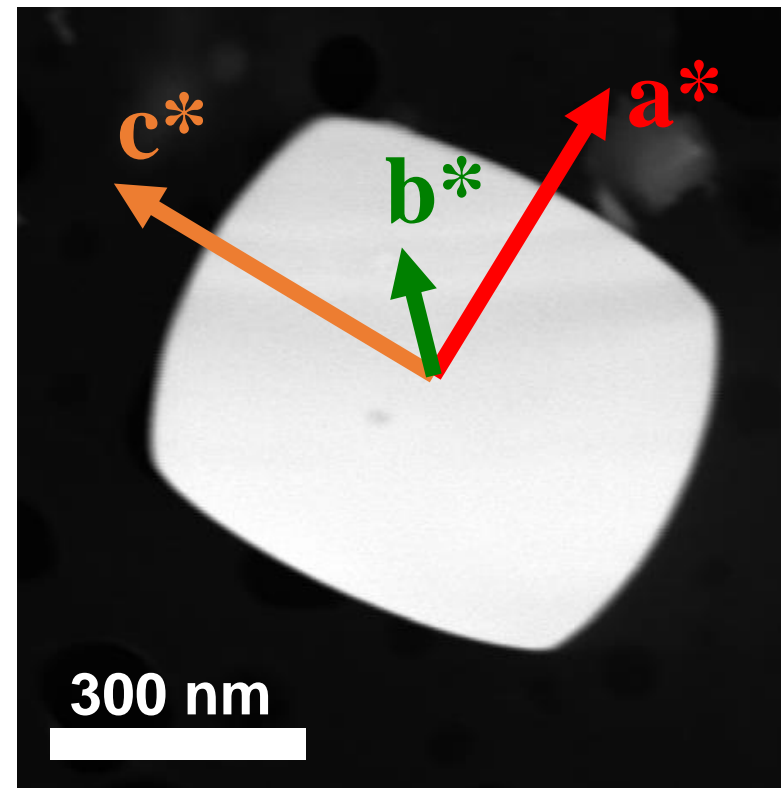


# Cell parameters & Orientation



**Cell parameters**

manual selection or clustering  
in difference vector space

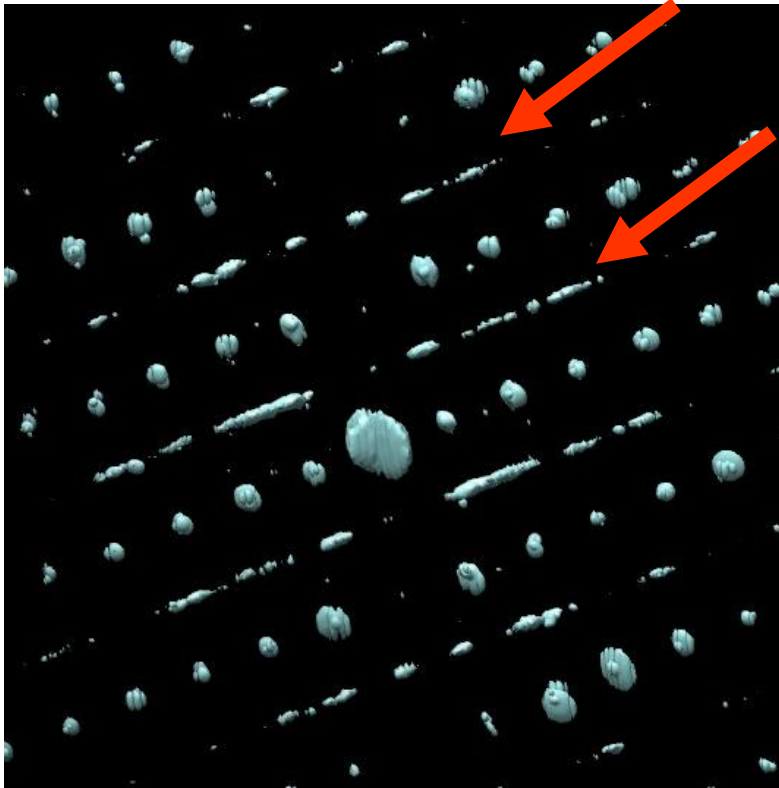


**Orientation matrix**

correlation with crystal shape for  
determination of direction of  
growth and facets

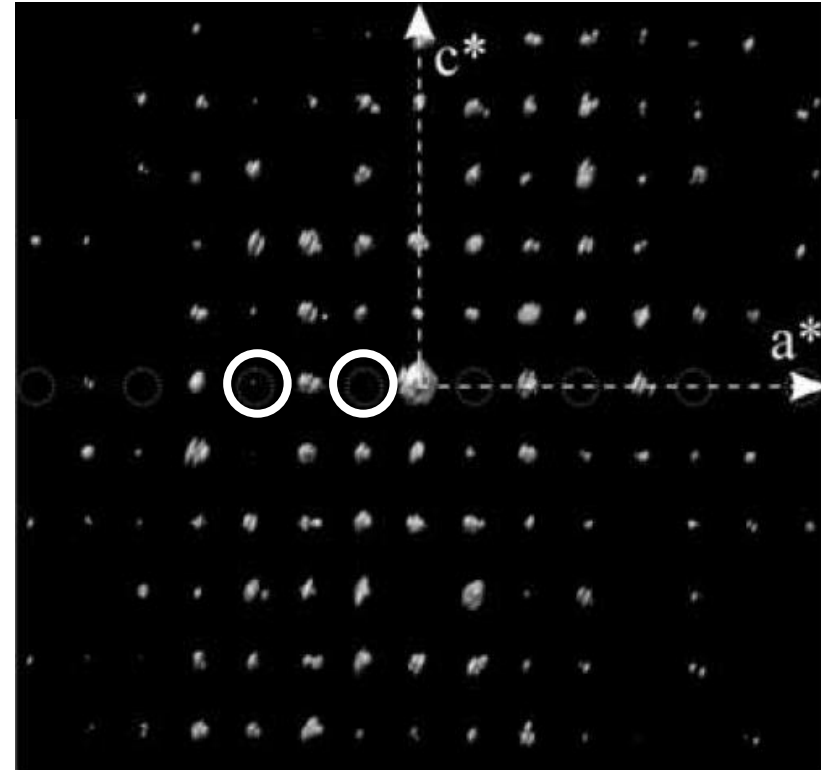


# Disorder & Symmetry



**Disorder**

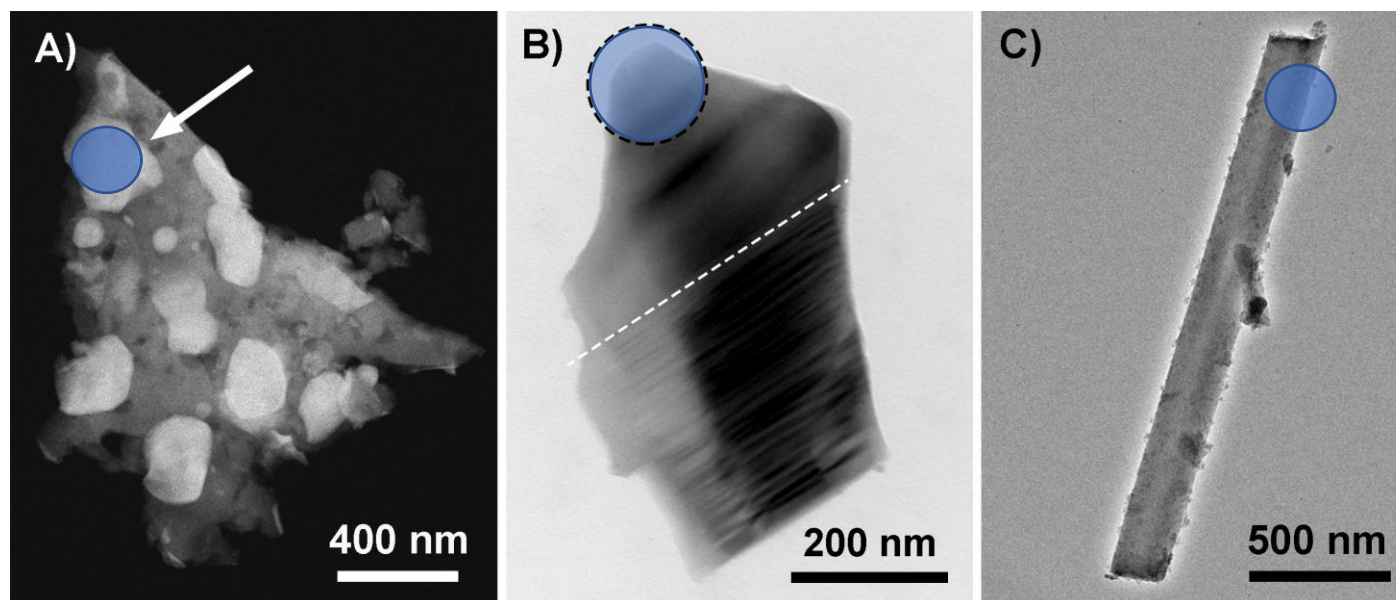
$$0kl : k = 2n+1$$



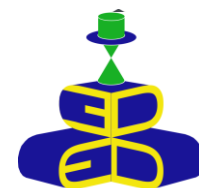
**Extinctions**

$$hk0 : h = 2n$$

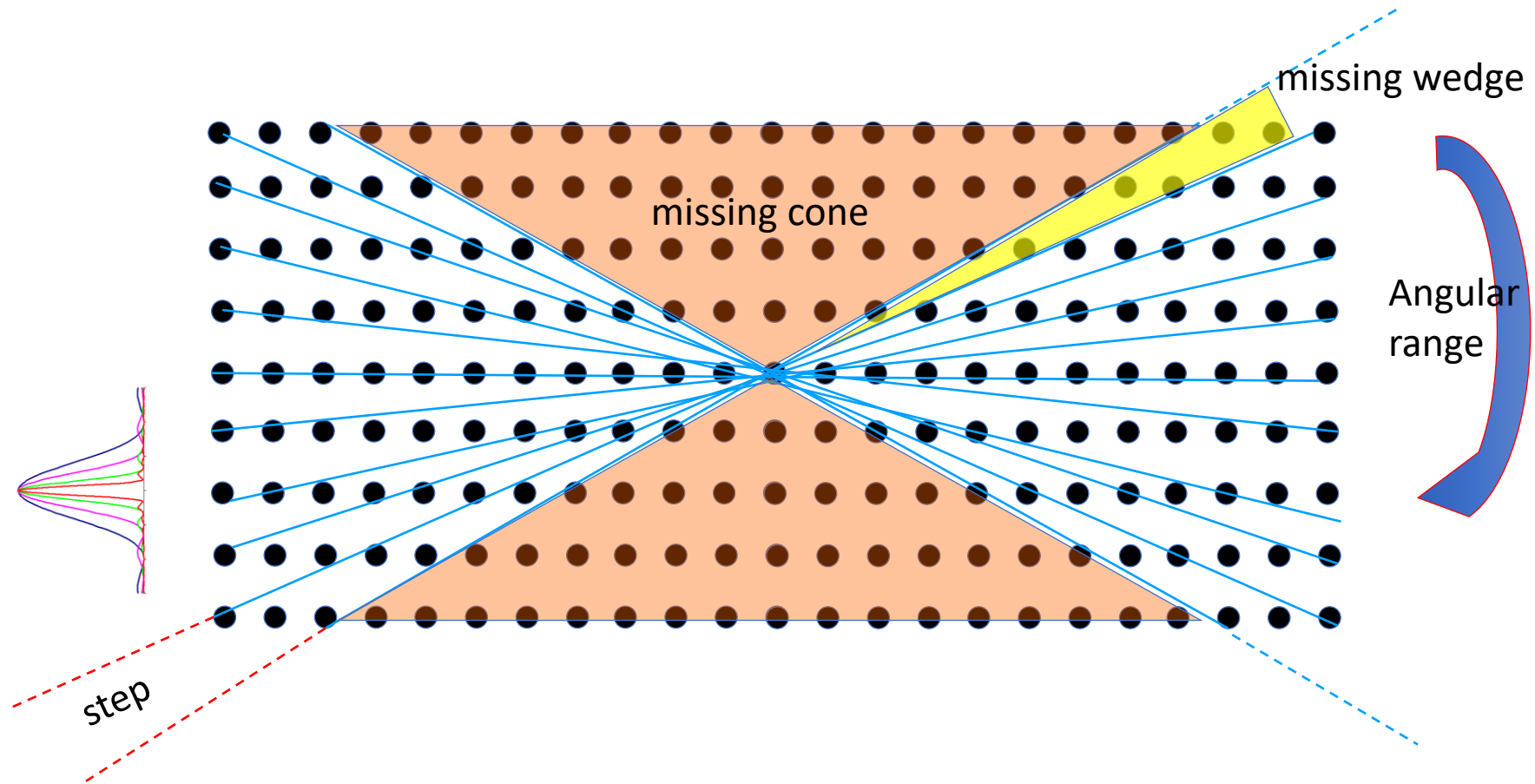
ALL THESE 3D INFO



FROM A VERY SMALL AREA



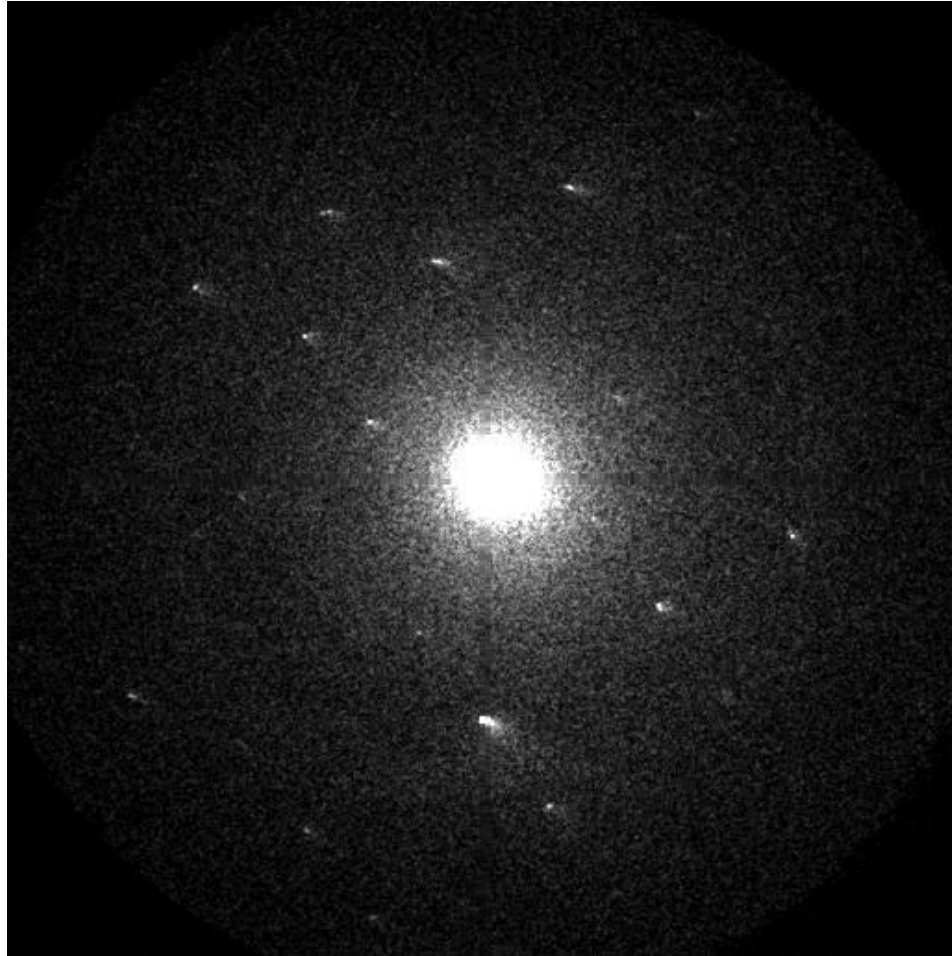
# ELECTRON DIFFRACTION TOMOGRAPHY



The missing wedge affects mainly the quality of the intensities:

Structure solution possible but not guarantee.....



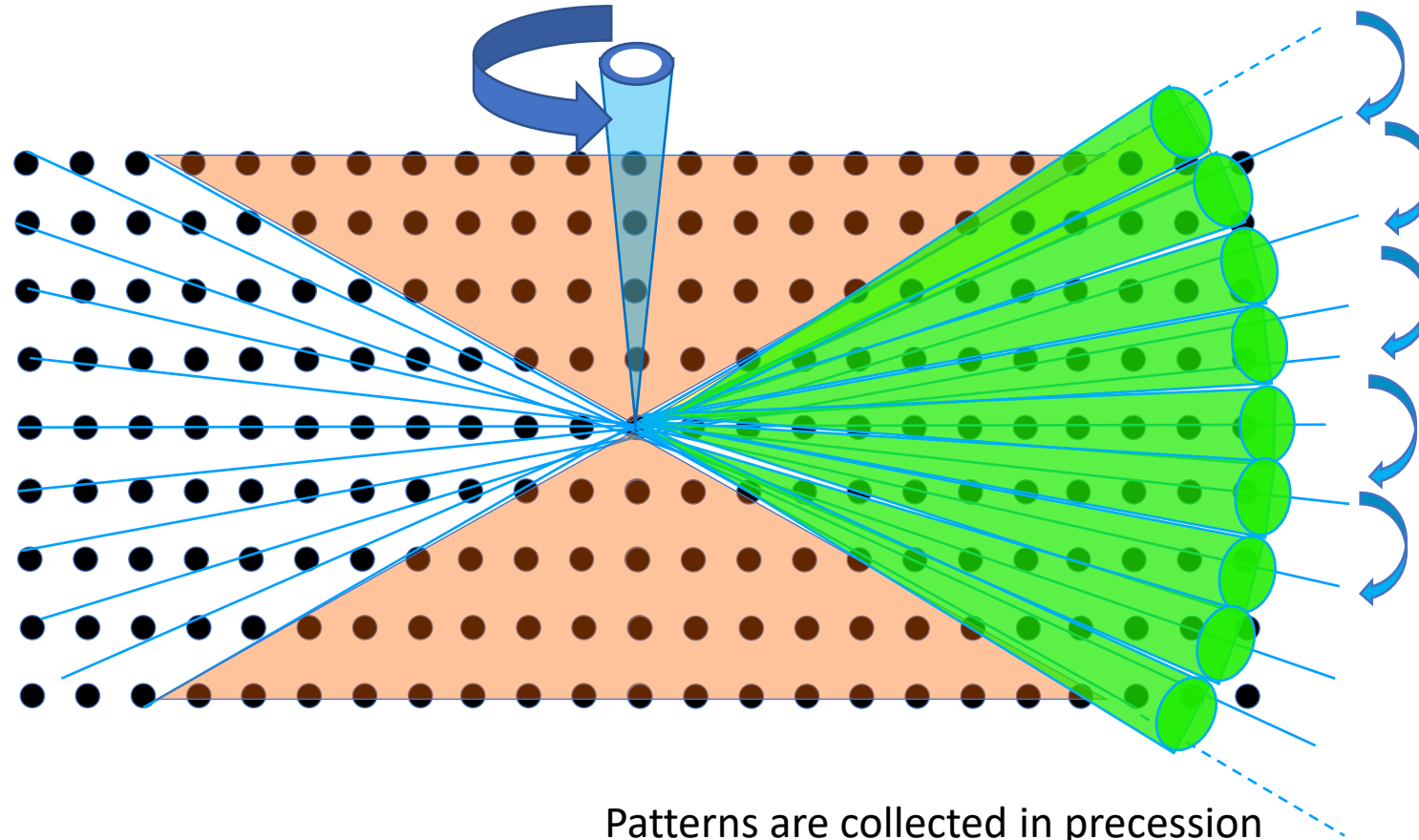


Angular step  $0.15^\circ$





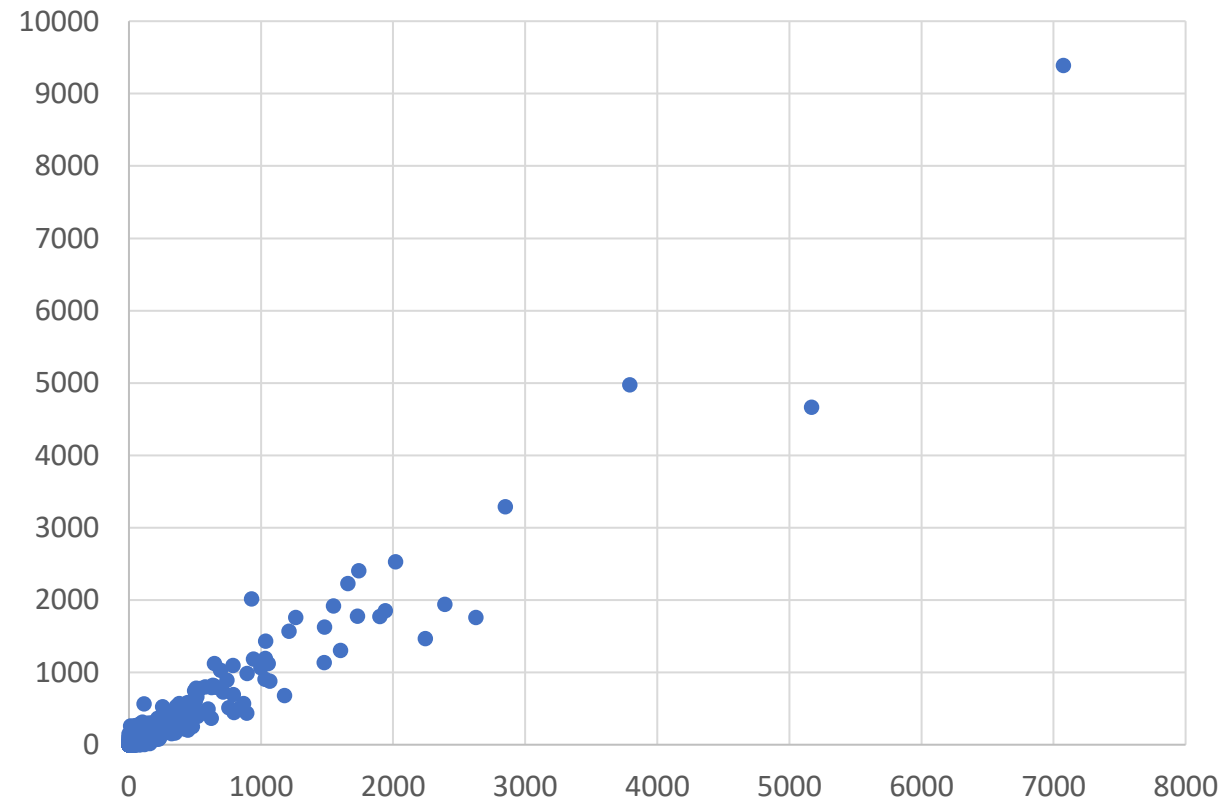
# PRECESSED ELECTRON DIFFRACTION TOMOGRAPHY PEDT



*Mugnaioli et al. Ultramicroscopy 109 (2009) 758*

Patterns are collected in precession  
mode with an aperture semiangle  
comparable with the angular step  
Angular step:  $1^\circ$   
Precession angle:  $1^\circ - 2^\circ$



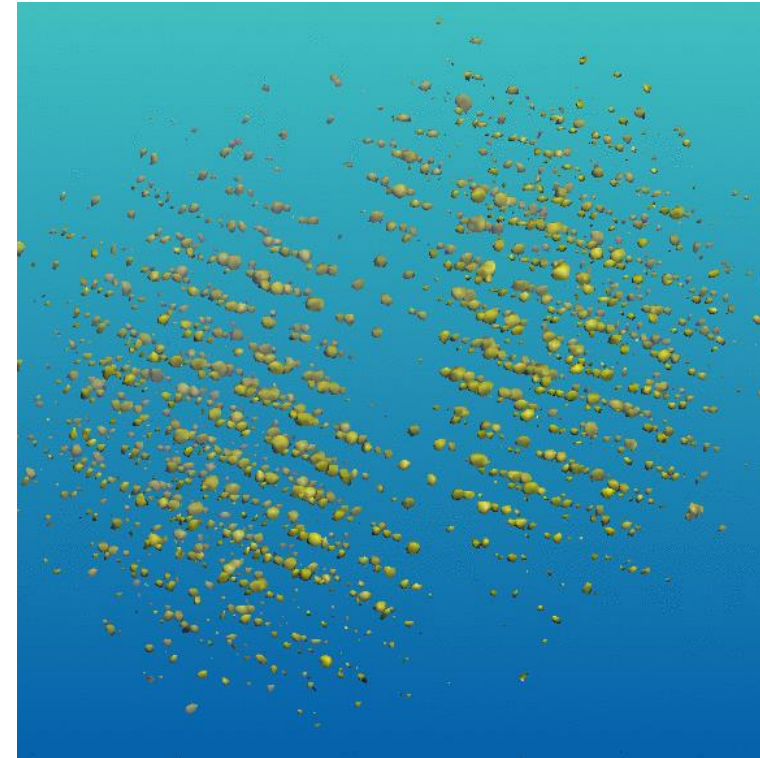
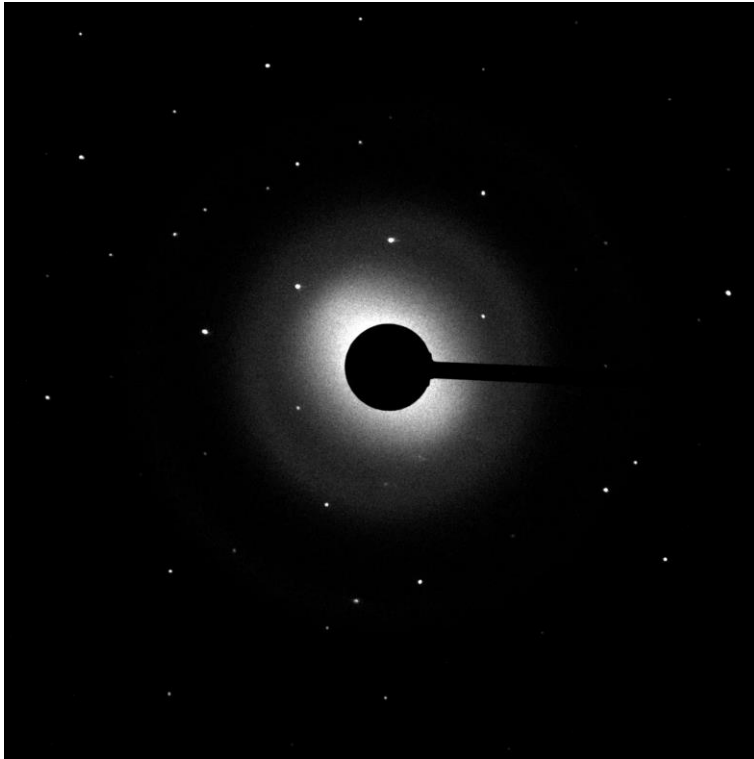


$F_{obs}$  vs  $F_{calc}$

- ❑ The intensity data are quasi-kinematical suitable for structure solution (direct methods, charge flipping,...), (SIR, SHELX, JANA)



## ELECTRON DIFFRACTION TOMOGRAPHY

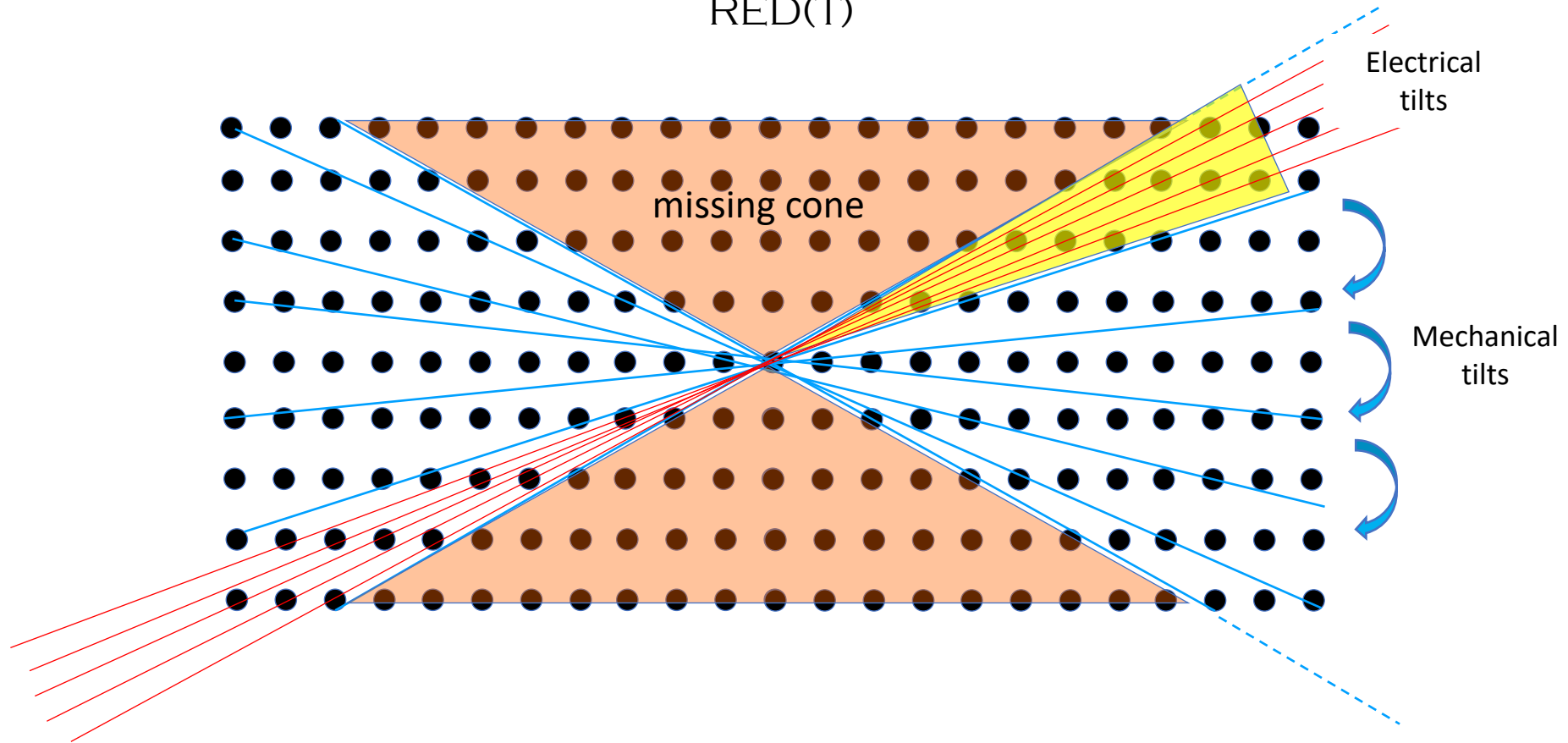


3D reconstruction of the  
reciprocal space

- Unit cell parameters, crystal symmetry
- From the reflection intensities get the crystal structure



# ROTATION ELECTRON DIFFRACTION TOMOGRAPHY RED(T)

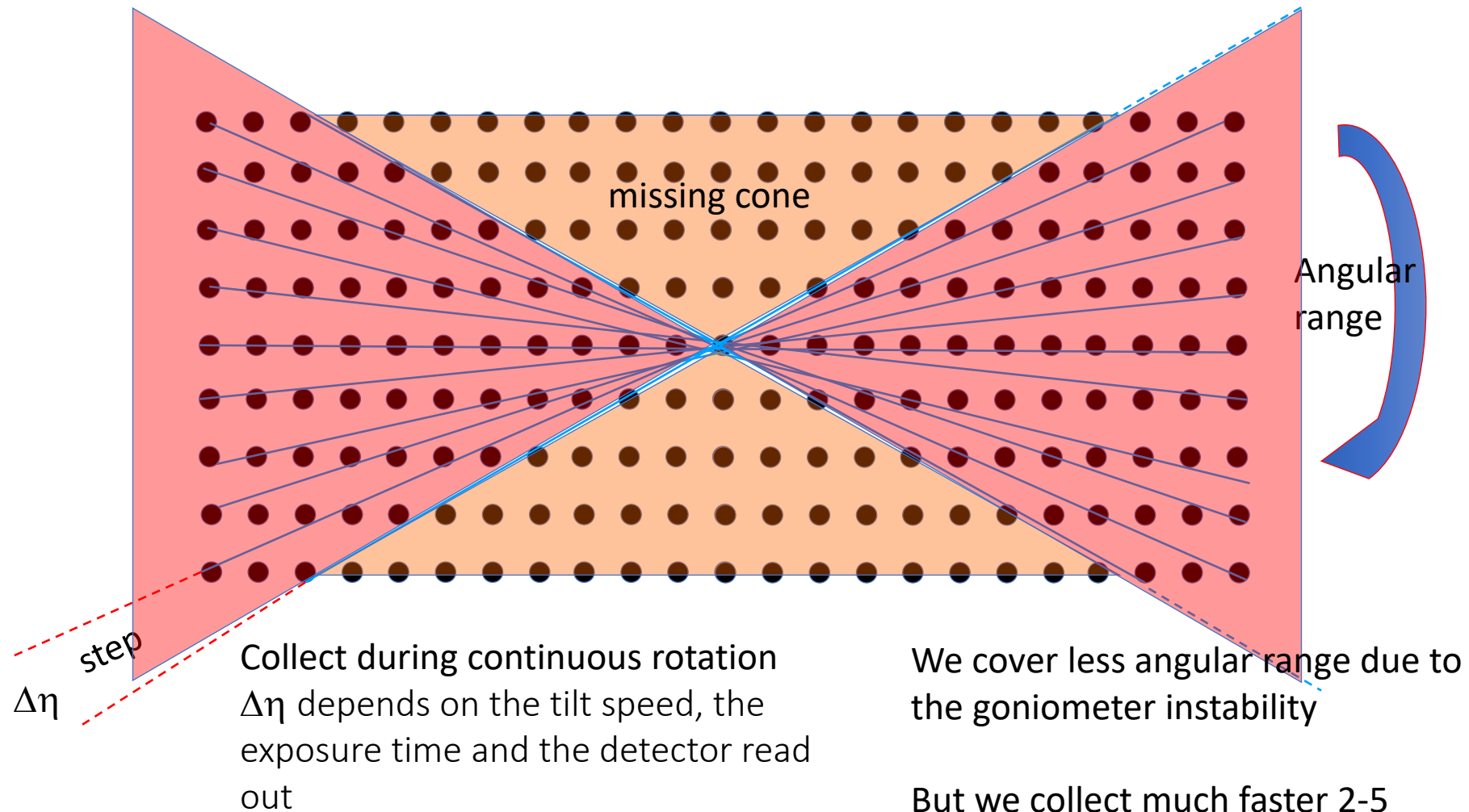


Zhang et al. Z. Kristallogr. 225 (2010) 94.





# FAST ELECTRON DIFFRACTION TOMOGRAPHY (CONTINUOUS ROTATION/MICROED)



*Neederlof et al. Acta Cryst. D69 (2013) 1223.*

*Nannenga et al. Nat. Methods 11 (2014) 927.*

*Gemmi et al. Journal of Applied Crystallography 48 (2015) 718.*



## For the continuous rotation we need a fast detector

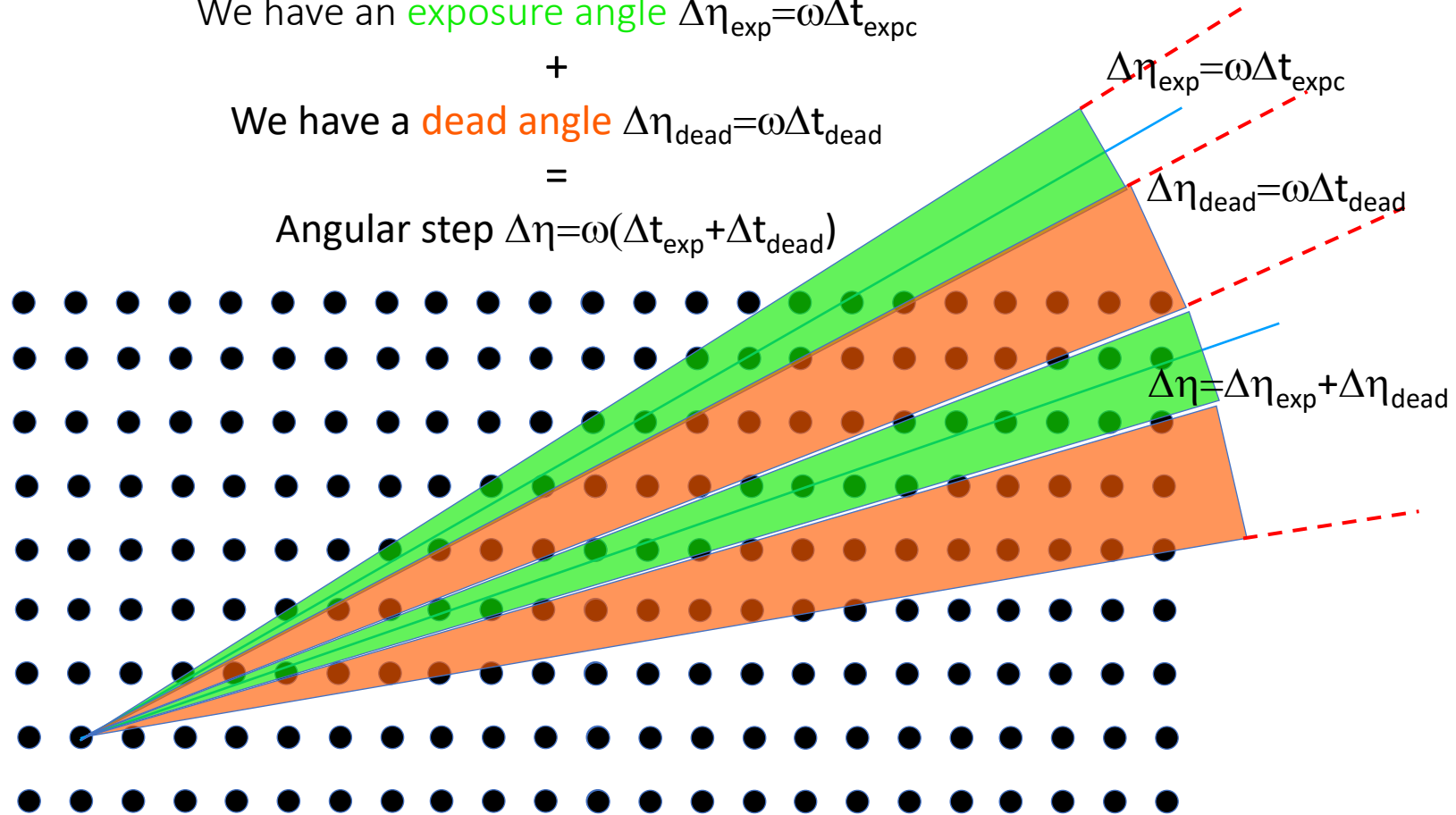
We have an **exposure angle**  $\Delta\eta_{\text{exp}} = \omega\Delta t_{\text{expc}}$

+

We have a **dead angle**  $\Delta\eta_{\text{dead}} = \omega\Delta t_{\text{dead}}$

=

Angular step  $\Delta\eta = \omega(\Delta t_{\text{exp}} + \Delta t_{\text{dead}})$



We can tune  $\Delta\eta$  by varying the rotation speed  $\omega$ : in our TEM from  $0.3^\circ$  to  $5^\circ$  per second

$\Delta\eta_{\text{dead}}$  must be minimized  $\rightarrow$  we need a fast detector

The crystal should not move during the continuous tilting



For the continuous rotation we need a fast detector

**Fast and sensitive detector:**

**MEDIPIX** single electron detection camera  
which has a read out time of 7 ms

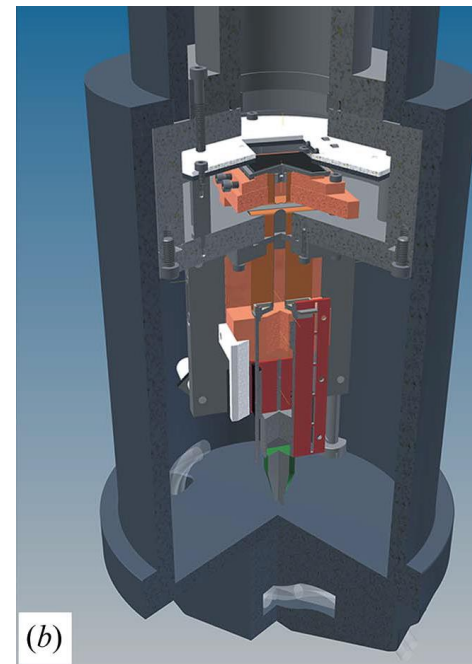
The fastest is the detector the fastest we can rotate  
the crystal the less time we illuminate the crystal:  
very low dose.

In our TEM we can vary the rotation speed in the interval

**$0.3^\circ/\text{s} - 5^\circ/\text{s}$**

With a dead time of 7 ms the lost reciprocal space angle  
varies between:

**$0.002^\circ$  and  $0.035^\circ$  negligible**



*Genderen et al. Acta Cryst (2016) A72, 236-242.*



# Continuous (fast) acquisition

Timepix detector  
512x512 px

Exposure: 0.45 s

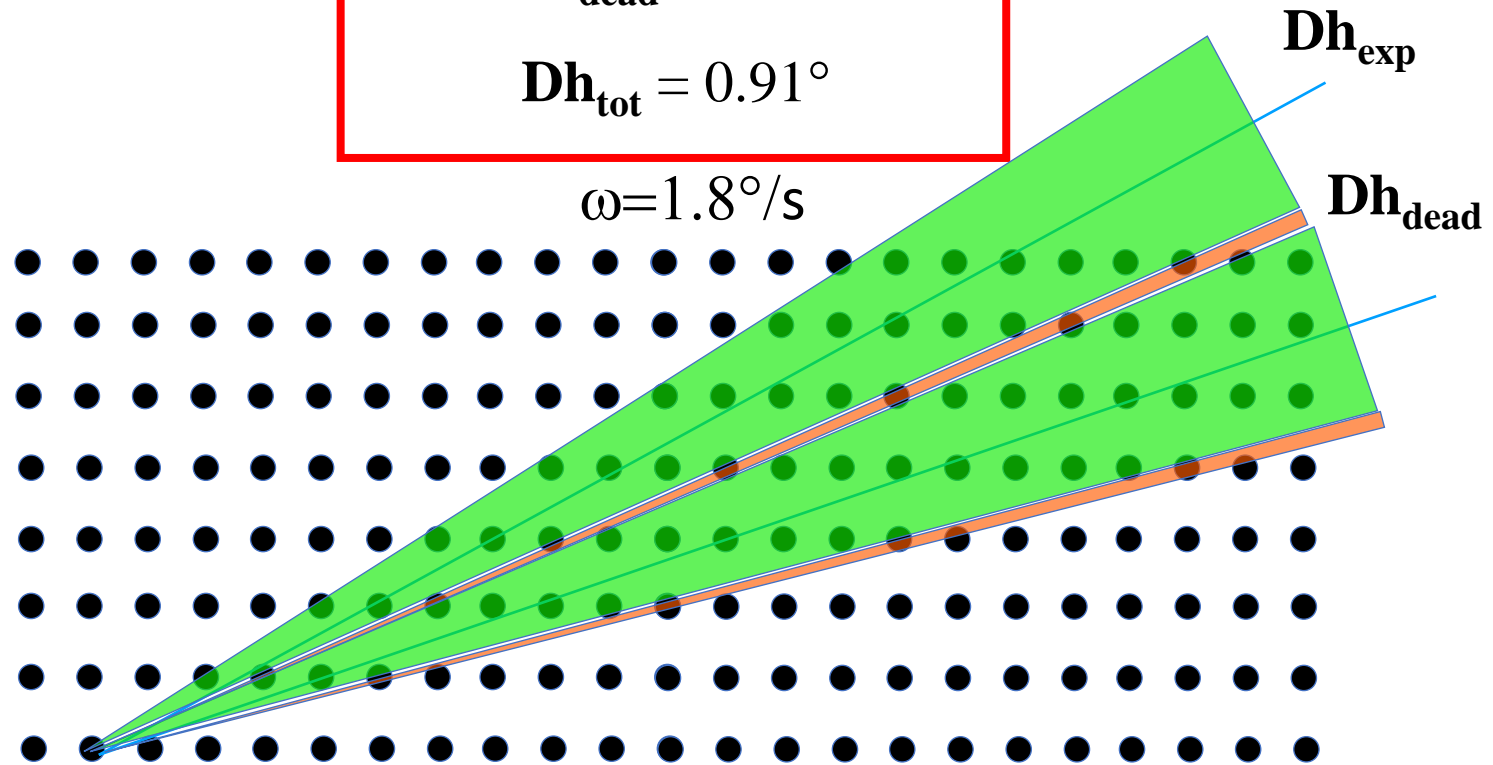
$$Dh_{\text{exp}} = 0.83^\circ$$

$$Dh_{\text{dead}} = 0.08^\circ$$

$$Dh_{\text{tot}} = 0.91^\circ$$

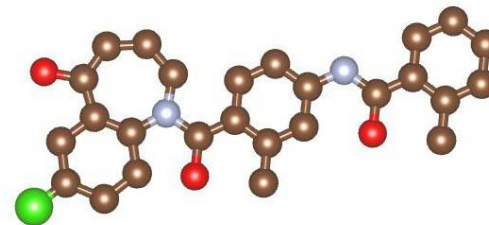
IEDT

$$\omega = 1.8^\circ/\text{s}$$





# Low Dose

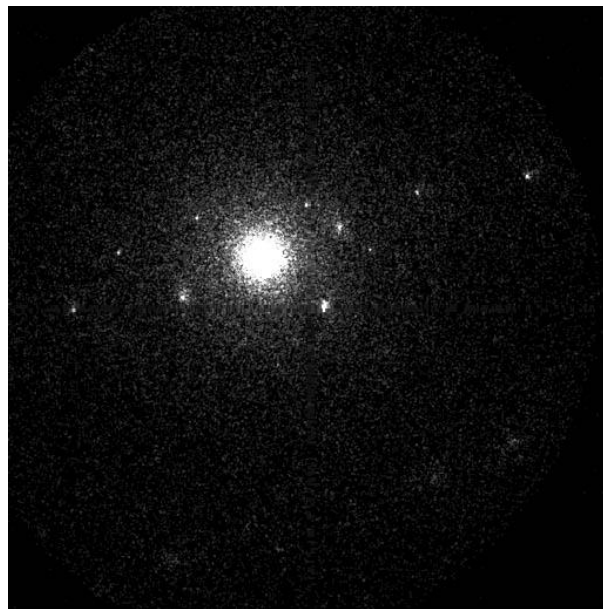


API1

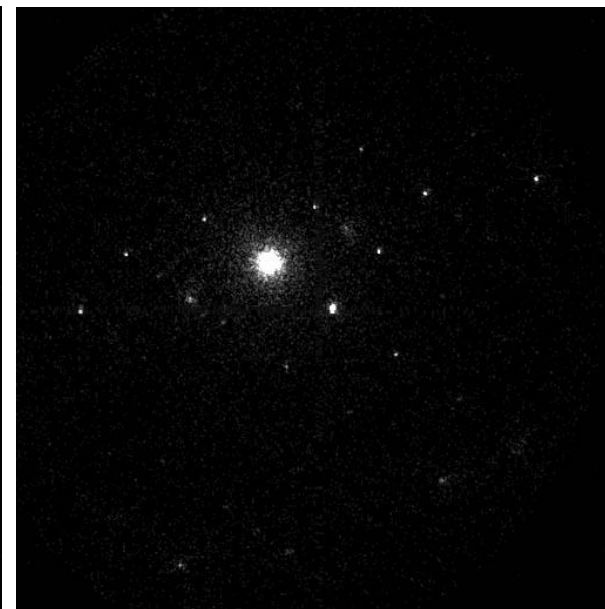
MEDIPIX



CCD + Energy Filter



MEDIPIX no Energy Filter

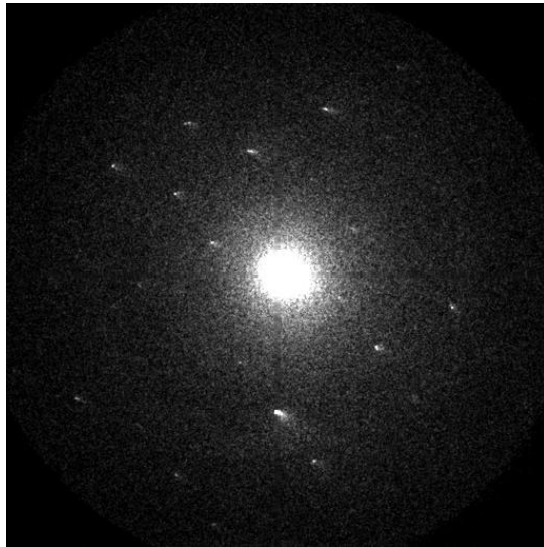


MEDIPIX + Energy Filter

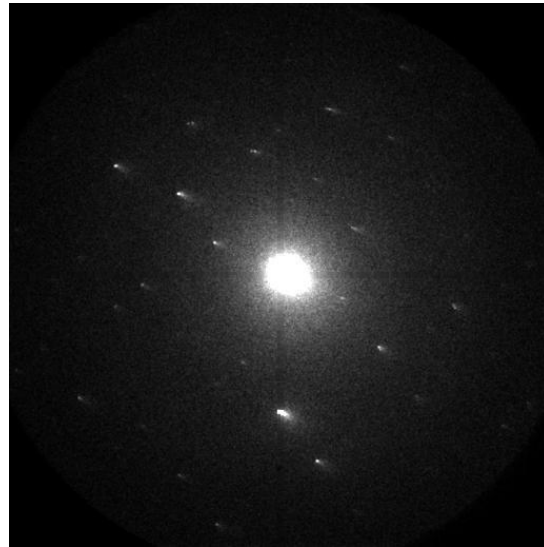
The dose can be reduced  $<0.05 \text{ e} \text{ \AA}^{-2} \text{ s}^{-1}$



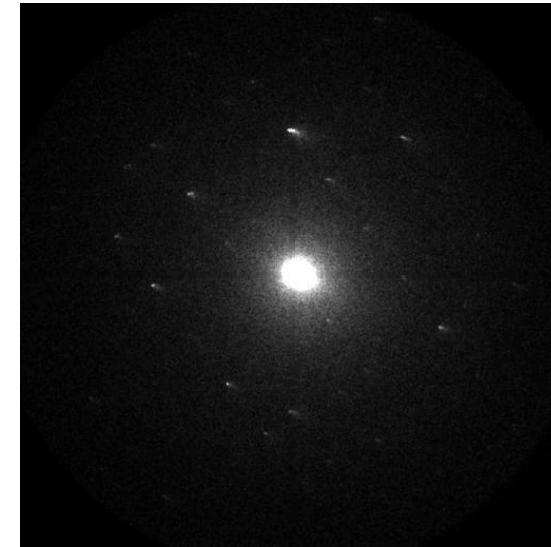
## FROM REDT TO IEDT



0.15°



0.50°

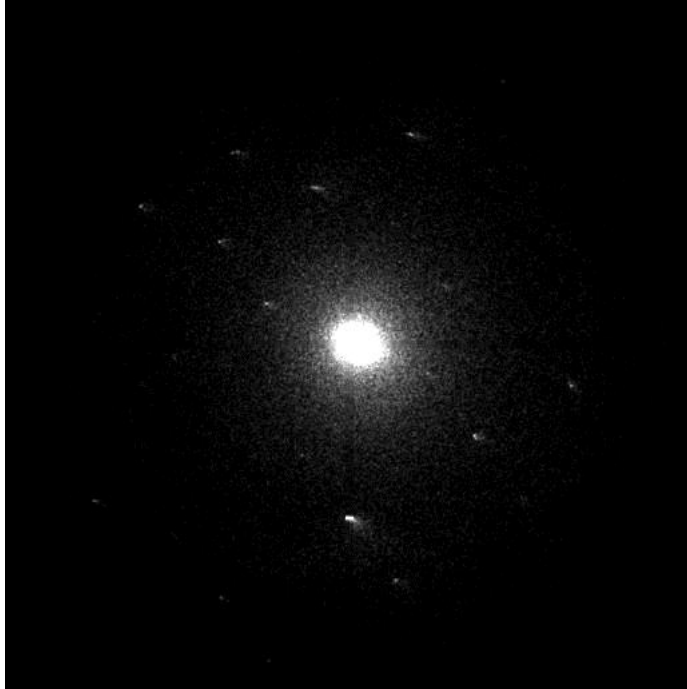


1°

Angular integration

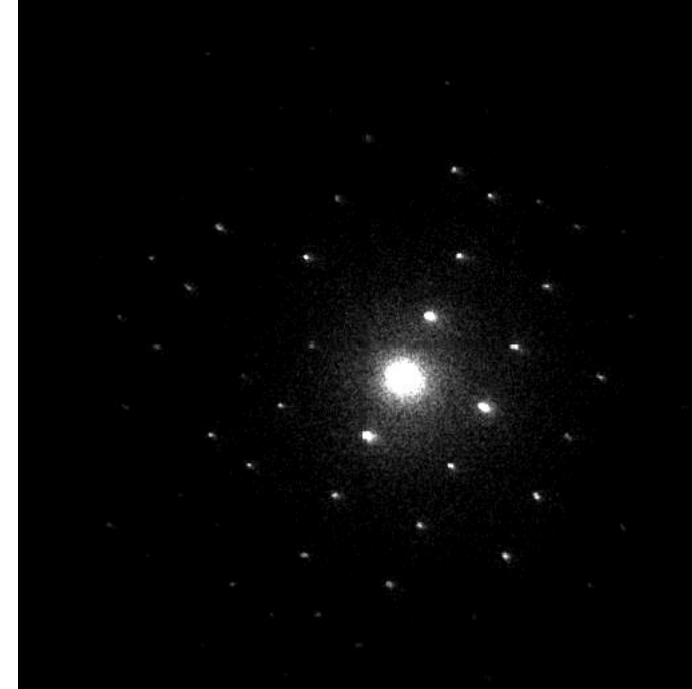


# Continuous vs stepwise



IEDT or cRED

0.15° each frame



PEDT stepwise

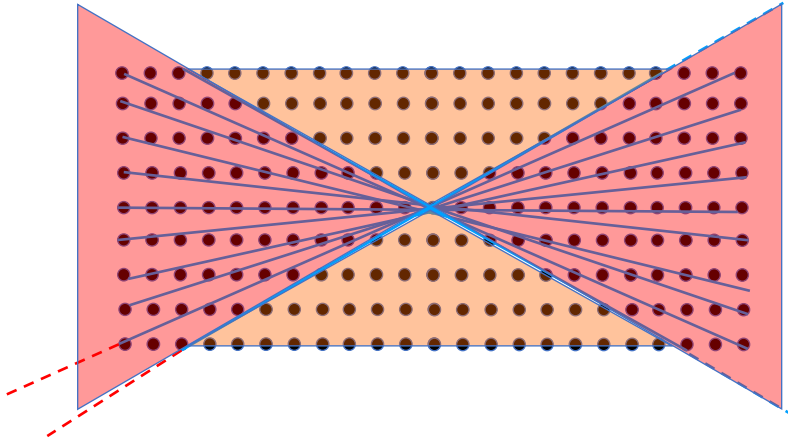
1° prec

1° each frame



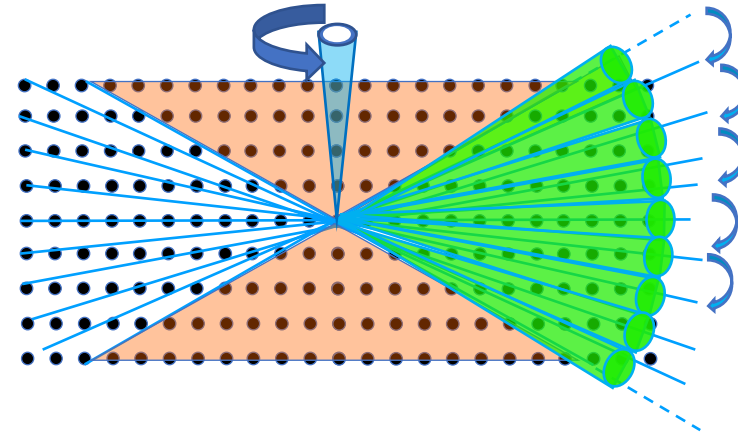
# 3D ED

Continuous



- You can cover smaller angular range of the reciprocal space  $40^{\circ}$ - $80^{\circ}$
- You can go faster 2-5 minutes, but also 30 seconds
- You can work in low dose

Step wise



- You can cover large angular range of the reciprocal space  $100^{\circ}$ - $120^{\circ}$
- it is time consuming 15 min- 30 min





# DIFFERENT 3D ED

Nano diffraction Precession mode	Mechanical 1°	Automatically STEM
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SAED Precession mode	Mechanical 1°	Semi automatic/ Manual TEM
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SAED Stationary mode	Mechanical 1°	Manually TEM
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SAED Stationary mode	Mechanical 2° Electrical 0.01°	Automatic TEM
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SAED or Nanodiff Stationary mode	Continuous	Goniometer Stability
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## Automatic PEDT

Kolb et al. Ultram. 107 (2007) 507  
Mugnaoli et al. Ultram. 109 (2009) 758

## “Manual” PEDT

Palatinus et al. Inorg.Chem. 50 (2011) 3743  
Gemmi et al. Acta Cryst B68 (2012) 15.

## “Manual” EDT no prec

Fan et al. Inorg. Chem. 52 (2013) 11067.

## Rotation

Singh et al. J. Appl. Cryst 47 (2014) 1  
Mayence et al. Inorg. Chem. 53 (2014) 5067.

## Continuous Rotation – IEDT - MICROED

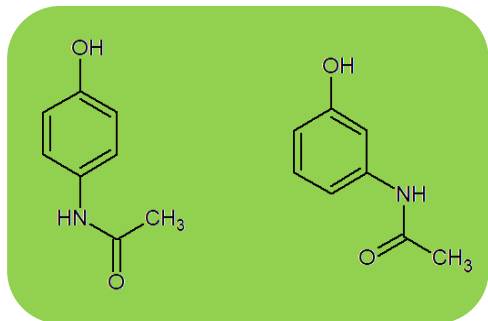
Neederlof et al. Acta Cryst. D69 (2013) 1223.  
Nannenga et al. Nat. Methods 11 (2014) 927.  
Gemmi et al. Journal of Applied Crystallography 48 (2015) 718.



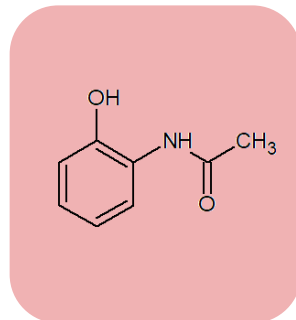
## ORTHOCETAMOL FROM VODES

### Orthocetamol

- ✓ demonstrates lower hepatotoxicity than widely used Paracetamol
- ✓ promising as an anti-arthritic treatment

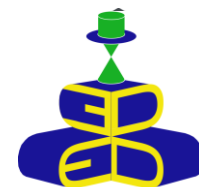
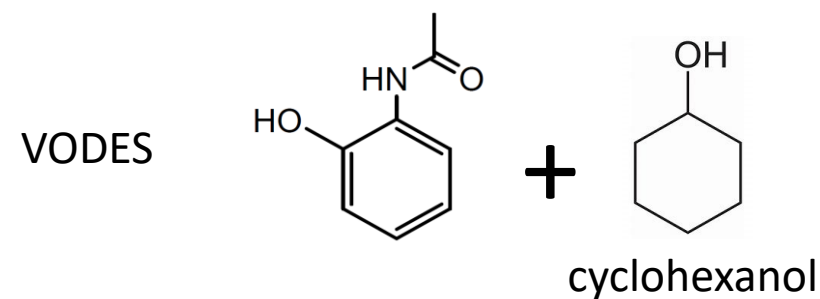
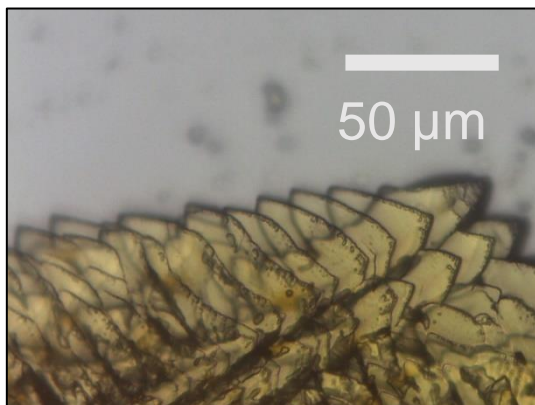
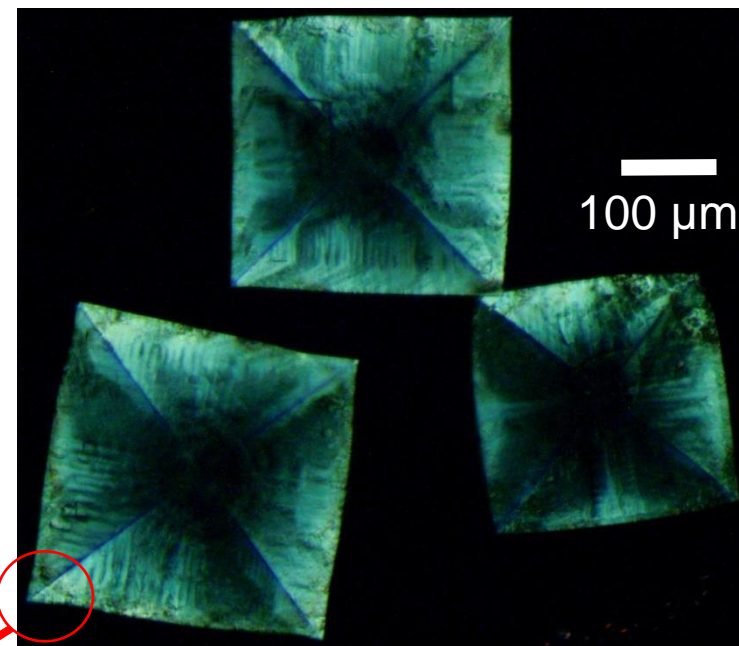


Paracetamol & Metacetamol are well known.

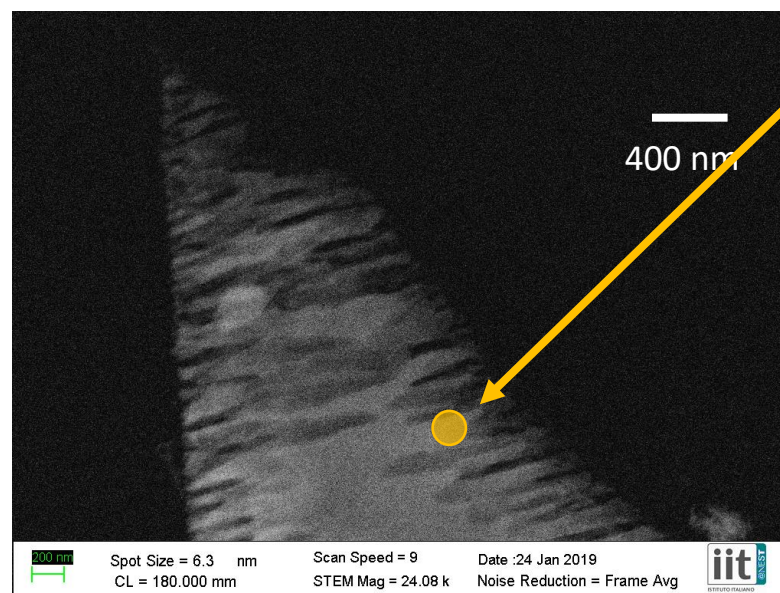
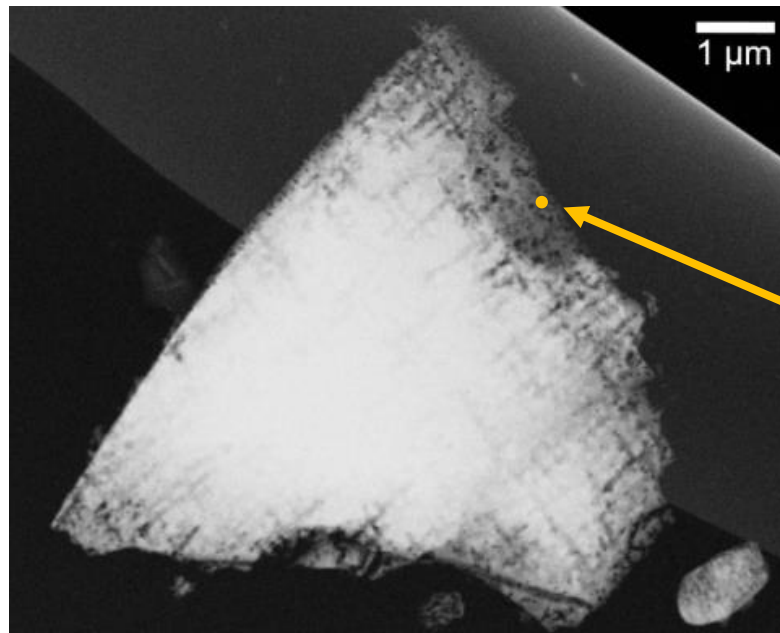


Orthocetamol was first reported in 1876, before the discovery of X-ray.

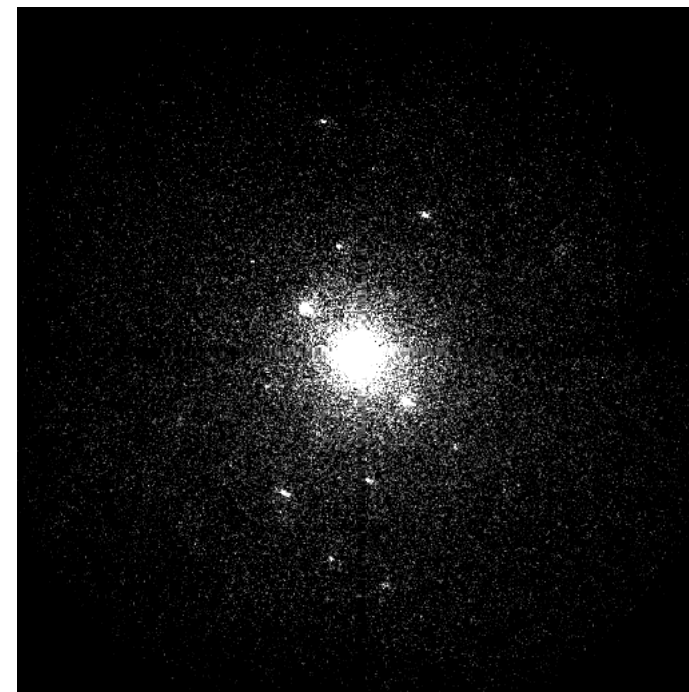
**Structure is still unknown!**



# ORTHOCETAMOL FROM VODES



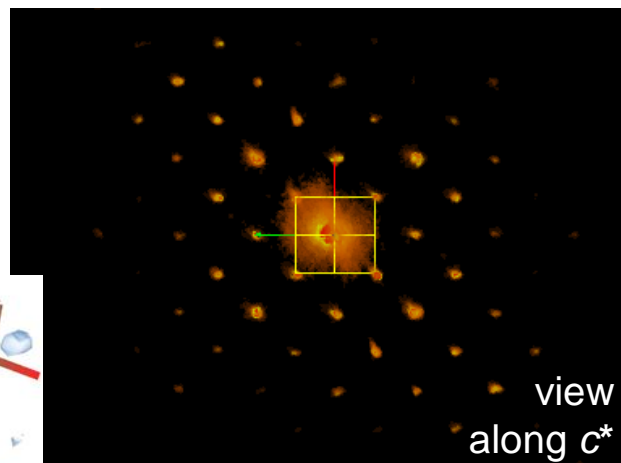
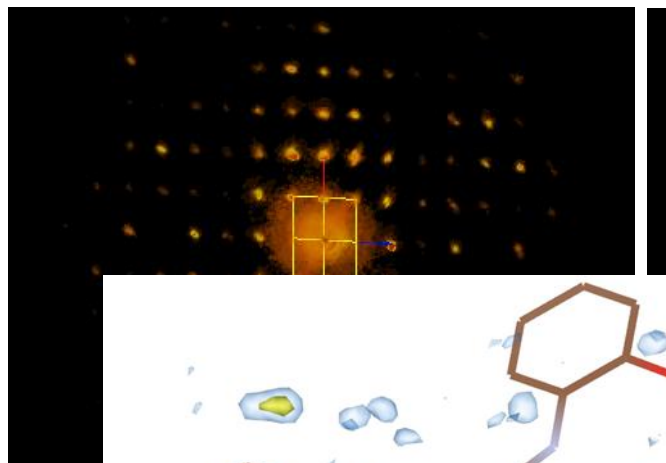
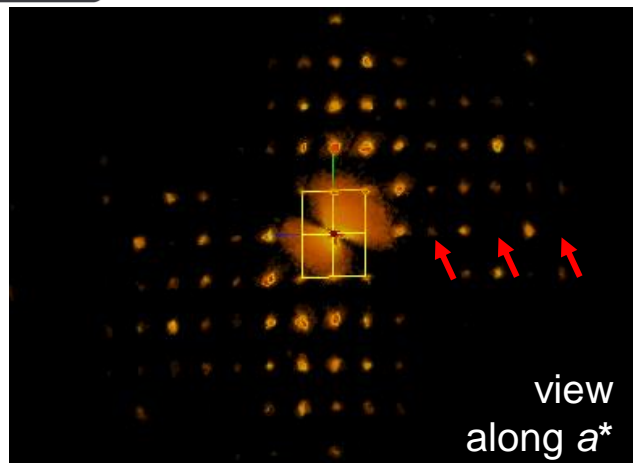
3D ED data collection in precession mode



Multiple twinning single domains of few hundreds of nanometers

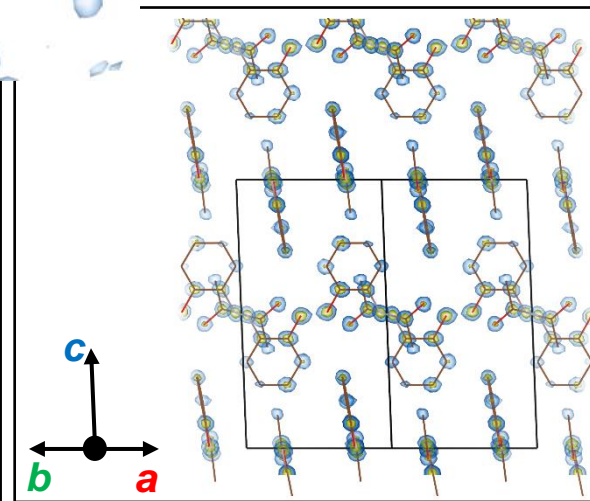
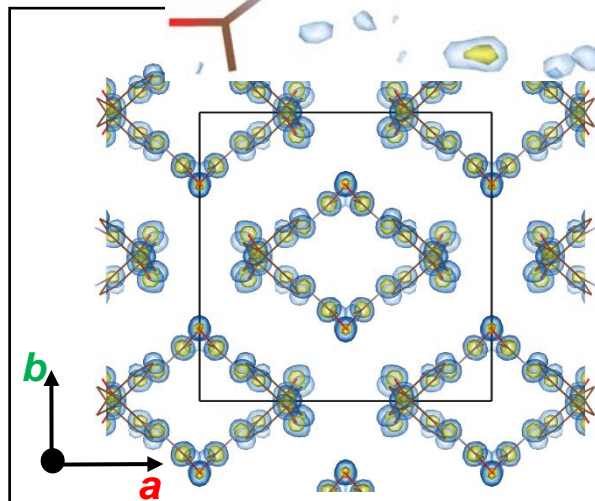
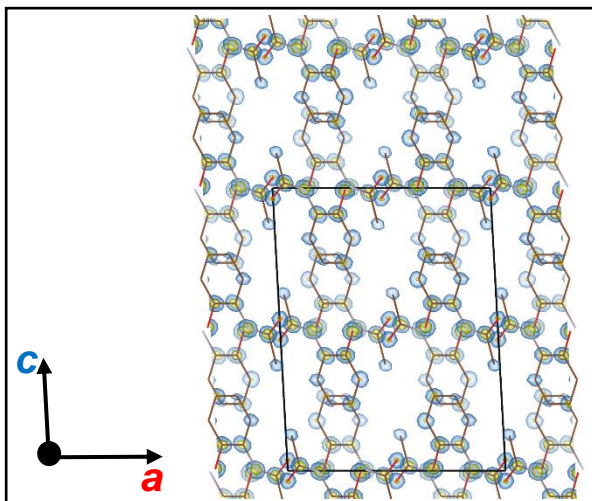


## ORTHOCETAMOL FROM VODES



$$\begin{aligned} a &= 10.56 \text{ \AA} \\ b &= 10.39 \text{ \AA} \\ c &= 13.72 \text{ \AA} \\ \beta &= 93.113^\circ \end{aligned}$$

**C-centered monoclinic  $C2/c$**   
(pseudo-tetragonal)



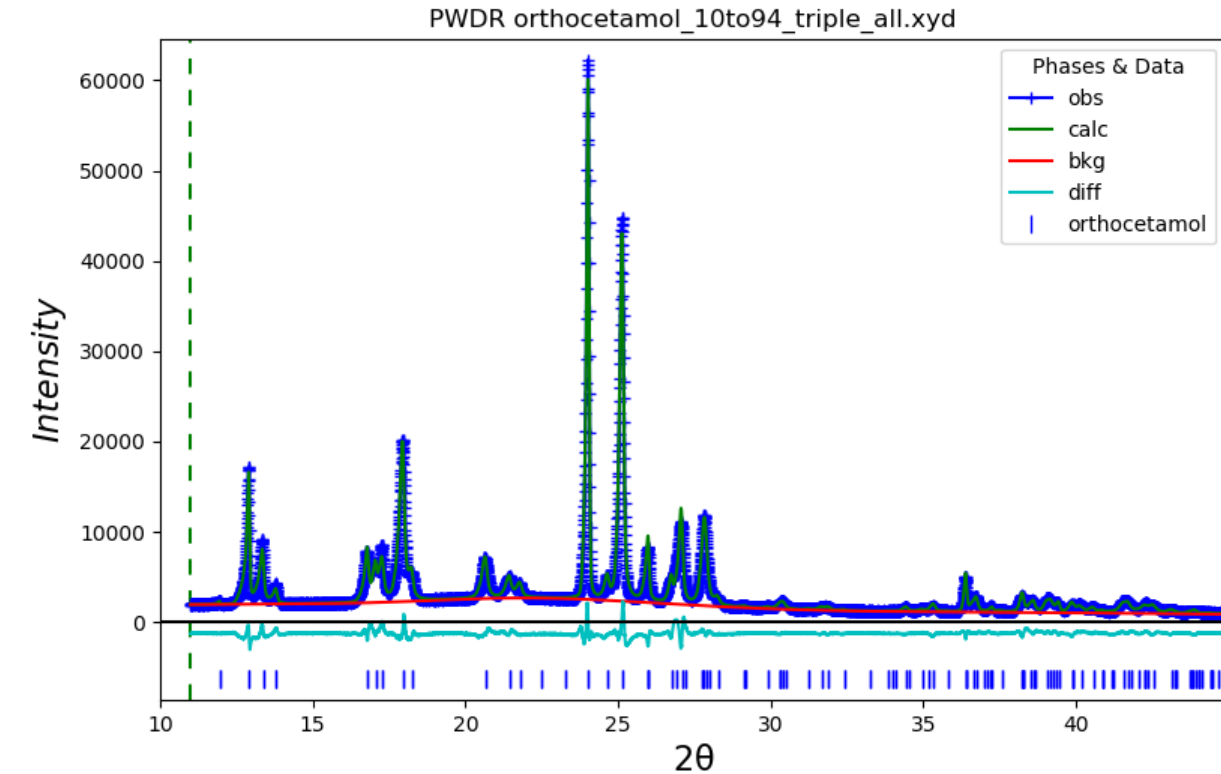
Solved ab-initio by direct  
methods in  **$C2/c$** .

All 11 non-hydrogen atoms  
were clearly recognizable.

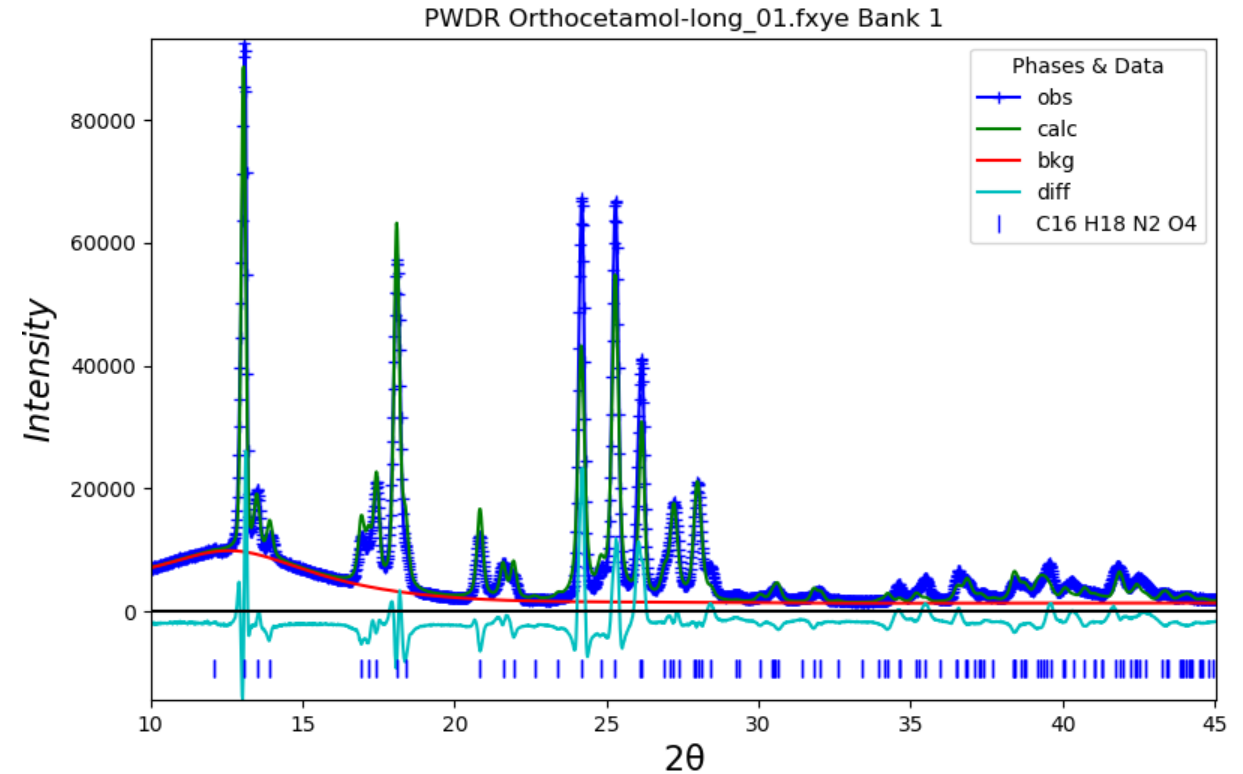




# ORTHO CETAMOL FROM VODES



Debye-Scherrer data collection



Bragg-Brentano

Although it's a pure phase, automatic indexing from PXRD was not successful.  
Problems: peak broadening, weak intensity above 30 ° 2theta (d= ca. 3 Å).





Thank you

